

Review

Microwave-Assisted Preparation of Luminescent Inorganic Materials: A Fast Route to Light Conversion and Storage Phosphors

José Miranda de Carvalho ^{1,*}, Cássio Cardoso Santos Pedroso ², Matheus Salgado de Nichile Saula ³,
Maria Claudia França Cunha Felinto ⁴ and Hermi Felinto de Brito ³

¹ Institute of Physics, University of São Paulo, São Paulo BR-05508-900, SP, Brazil

² The Molecular Foundry, Lawrence Berkeley National Laboratory, Berkeley, CA 94720, USA; ccspedroso@lbl.gov

³ Institute of Chemistry, University of São Paulo, São Paulo BR-05508-000, SP, Brazil; matheus.saula@usp.br (M.S.d.N.S.); hefbrito@iq.usp.br (H.F.d.B.)

⁴ Nuclear and Energy Research Institute, São Paulo BR-05508-000, SP, Brazil; mfelinto@ipen.br

* Correspondence: jmc@iq.usp.br

Abstract: Luminescent inorganic materials are used in several technological applications such as light-emitting displays, white LEDs for illumination, bioimaging, and photodynamic therapy. Usually, inorganic phosphors (e.g., complex oxides, silicates) need high temperatures and, in some cases, specific atmospheres to be formed or to obtain a homogeneous composition. Low ionic diffusion and high melting points of the precursors lead to long processing times in these solid-state syntheses with a cost in energy consumption when conventional heating methods are applied. Microwave-assisted synthesis relies on selective, volumetric heating attributed to the electromagnetic radiation interaction with the matter. The microwave heating allows for rapid heating rates and small temperature gradients yielding homogeneous, well-formed materials swiftly. Luminescent inorganic materials can benefit significantly from the microwave-assisted synthesis for high homogeneity, diverse morphology, and rapid screening of different compositions. The rapid screening allows for fast material investigation, whereas the benefits of enhanced homogeneity include improvement in the optical properties such as quantum yields and storage capacity.

Keywords: microwave-assisted synthesis; luminescent inorganic materials; light converting; energy storage; microwave heating



Citation: Miranda de Carvalho, J.; Pedroso, C.C.S.; Saula, M.S.d.N.; Felinto, M.C.F.C.; Brito, H.F.d. Microwave-Assisted Preparation of Luminescent Inorganic Materials: A Fast Route to Light Conversion and Storage Phosphors. *Molecules* **2021**, *26*, 2882. <https://doi.org/10.3390/molecules26102882>

Academic Editor: Farid Chemat

Received: 26 March 2021

Accepted: 9 May 2021

Published: 13 May 2021

Publisher's Note: MDPI stays neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Copyright: © 2021 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

1. Introduction

1.1. Luminescent Materials: Optical Properties and Applications

Luminescent materials, also known as phosphors, usually refers to inorganic materials that emit light after external physical stimuli, such as UV-VIS, IR-Lasers, X-rays, γ -rays, and e-beams [1]. These materials are composed of a host lattice with intentionally added optically active impurities (dopants). The dopants act as luminescent centers, converting the incident excitation into a characteristic light emission that depends on the dopant's identity [2]. The most common dopants for luminescent materials are the f-metals, such as the trivalent rare-earths (RE^{3+} : Pr, Nd, Eu, Tb, Dy, Er), that emit light after optical transitions within the f-electron manifold [2]. The f-electrons are strongly shielded from the chemical environment and retain atomic characteristics, yielding sharp emission lines with wavelengths nearly independent of the host's composition. The f-f electronic transitions are parity forbidden and, thus, the emission decay times are long (μ s-ms).

Some rare earth ions (e.g., Ce^{3+} , and Eu^{2+}) behave as crystal-field sensitive ions, having intense broad emission bands with emission color dependent on the host's composition. In these cases, the emission is governed by parity-allowed $4f-5d$ transitions, yielding intense

emission bands with fast decay times (ns- μ s). Figure 1 provides the comparison between the $f-f$ and $f-d$ transitions for the Eu^{3+} and Eu^{2+} ions.

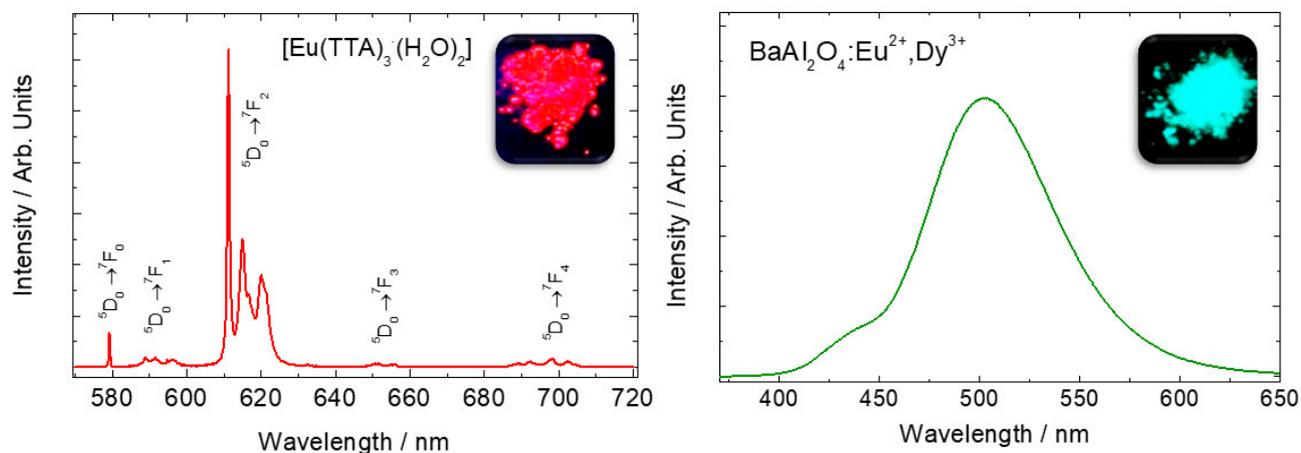


Figure 1. Emission spectra of Eu^{3+} in $\text{Eu}(\text{TTA})_3 \cdot (\text{H}_2\text{O})_2$ complex (left), and Eu^{2+} -doped BaAl_2O_4 host (right). Eu^{3+} sharp lines spectra is attributed to $f-f$ transitions whereas the Eu^{2+} broad emission band is attributed to $f-d$ transitions. Adapted from references [3,4]. Copyright © 1997 Published by Elsevier B.V.

Various transition metals also act as luminescent centers, e.g., Mn^{4+} -doped K_2SiF_6 [5], and Cr^{3+} -doped ZnGa_2O_4 [6]. In these cases, parity-forbidden $d-d$ electronic transition is responsible for the emission. The d -orbitals broadly interact with the chemical environment leading to strong crystal field splitting according to the composition of the materials. Thus, most of the transition metals' absorption and emission bands are broad. However, sharp emissions can also be observed, such as Mn^{4+} -doped K_3ScF_6 red phosphors [7].

Luminescent materials can act as light converters due to different photophysical processes, such as downconversion (or quantum cutting) [8–10], upconversion [11–13], and down-shifting [14,15] (Figure 2A–C). Downshifting is the most conventional photoluminescence mechanism, where high-energy photons are converted into low-energy photons. Ideally, all excited photons are converted to emitted photons without losses in the optical process, obtaining a quantum yield of 100% [16–19]. On the other hand, the first experimental evidence for quantum yields higher than 100% was reported for $\text{YF}_3:\text{Pr}^{3+}$ material [17,20]. This optical phenomenon proposed by Dexter consists of the high-energy photons splitting into two or more lower-energy photons. This mechanism, called downconversion or quantum cutting, involves single or multiple ions (e.g., in $\text{YF}_3:\text{Pr}^{3+}$ [17,20] or $\text{YPO}_4:\text{Yb}^{3+}$, Tb^{3+} [10], respectively).

Upconversion (UC) is a nonlinear optical phenomenon that converts two or more low-energy photons to a higher-energy photon [13]. This optical process, known as anti-Stokes emission, was theoretically proposed by the physicist Nicolaas Bloembergen in 1959 [21]. The first experimental result, reported by François Auzel in 1966, consisted of the energy transfer from Yb^{3+} to Er^{3+} and Tm^{3+} ions [13]. The UC process has since inspired light conversion mainly from near-infrared (NIR) to UV-visible-NIR light, suitable for bioimaging, solar cells, anticounterfeiting, photocatalysis, and photodynamic therapy [22].

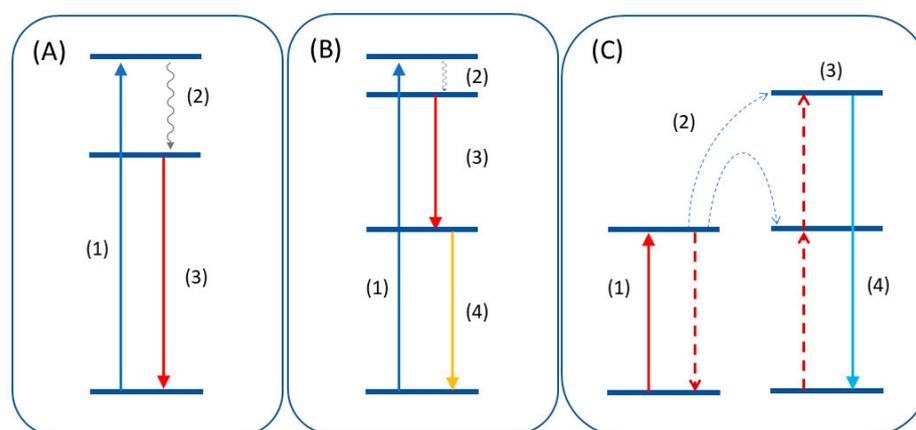


Figure 2. (A) Downshift mechanism consisting of excitation (1), non-radiative decay (2), and emission (3). (B) Quantum cutting mechanism where the emission of two photons (3,4) occurs after the excitation. (C) Upconversion emission after energy transfer (2) of two lower energy quanta to a higher excited state (3).

Persistent luminescence (PeL) materials continue to emit light after removing the excitation source due to stored energy in the structural defects that act as charge carrier traps [23–25]. After the removal of the excitation, the material absorbs available thermal (kT) or optical (hν) energy to release the trapped charge carriers that recombine with the emitting centers. The radiative decay of the excited states emits a photon with a characteristic wavelength. The most efficient PeL phosphors are dominated by Eu^{2+} -doped inorganic materials, exhibiting 24 to 48 h of persistent emission after few minutes of excitation in the UV-VIS range [25]. However, plenty of examples for d- and f-metals are available, and extensive reviews on the topic have been published [23,24].

Luminescent materials have several applications in technology, telecommunications, and health sciences, and are the subject of extensive research. Specific applications can be in light-emitting diodes [26–28], laser [29], thermometry [30], thermoluminescence dating [31,32], dosimetry [33], bioimaging [6,34], and photodynamic therapy [35–37].

For instance, phosphor converters for LEDs (pc-LEDs) work as light converters where an excitation source, e.g., blue emitting-LEDs, excites the luminescent material that emits broadband in the yellow region. This kind of phosphor has defined characteristics, such as high quantum yields and fast excited-state decay times. The most efficient light converters for pc-LED are inorganic hosts doped mainly with Ce^{3+} and Eu^{2+} [27,38–40]. The parity-allowed (f–d) electronic transition in Ce^{3+} and Eu^{2+} ions results in high-intensity emissions with short excited-state lifetimes (μs–ns). However, other examples can be easily found in trivalent rare earths, Eu^{3+} , Tb^{3+} , Pr^{3+} , Dy^{3+} , and d-metals (Mn^{4+}) doped materials [5,7,41–44]. PeL materials are in evidence nowadays due to vast and unexplored bioimaging applications [37], theranostics [45], and photocatalysis [46]. When it comes to application, persistent phosphors must have high quantum yield and high storage capacity [47,48].

1.2. Concepts and Mechanisms of Microwave Heating

Microwaves (MW) are electromagnetic radiation in the frequency interval of 300 MHz (1.24 μeV; 1 m) to 300 GHz (1.24 meV; 1 mm). Radiation in this range is mainly used in wireless telecommunication protocols, such as Bluetooth and wi-fi modems [49]. A narrow part of the microwave spectrum is used for industrial, domestic, scientific, and medical purposes (ISM bands) [50]. The industrial and domestic uses of microwaves come from the ability to heat certain materials depending on their dielectric properties [49,51]. For instance, the radiation in a domestic microwave oven (2.45 GHz; λ: 12.2 cm) interacts with the water contained in the food, heating it swiftly due to the intrinsic dielectric properties of the water molecules. Further, a precise tuning of microwave radiation parameters enables

microwave heating of different materials, such as metal oxides [52,53], carbon [49,54,55], and organic solvents [56].

Microwave heating properties differ significantly from conventional heating ones. In conventional heating, the external source, such as a resistive furnace, heats the material's surface, leading to a large temperature gradient compared to the material's core. Instead, dielectric heating occurs volumetrically, meaning that the material irradiated by the microwaves is heated locally, leading to a small temperature gradient [56] (Figure 3). Microwave-assisted heating rates are drastically steeper than conventional heating because of the absence of heat transfer from an external source.

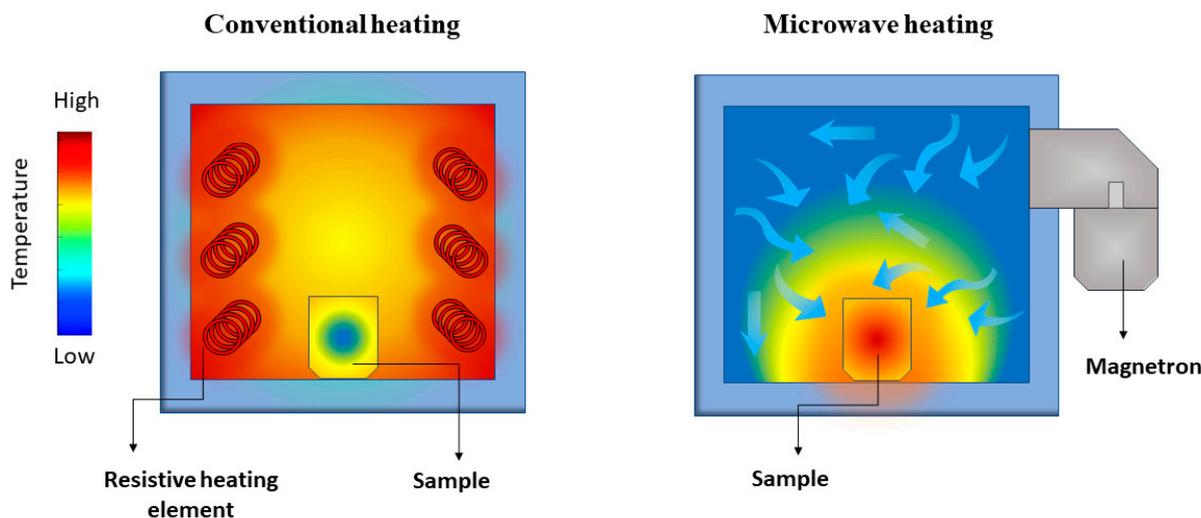


Figure 3. Representation of the temperature gradient in conventional (**left**) and microwave-assisted (**right**) heating of a solid sample.

The combination of these properties can be highly beneficial to the synthesis of inorganic materials using microwave-assisted approaches. For instance, solid-state methods benefit significantly from dielectric heating, shortening the synthesis time from days and hours to tens of minutes. In addition, it can yield more uniform, homogeneous materials [57].

The principles of microwave heating are based on the material's response to the electromagnetic fields' alternating nature. Bhattacharya et al. [49] discussed in detail the microwave heating mechanism in several materials. The relevant parameters for the heating include the dielectric constant (ϵ') and dielectric loss (ϵ''). The parameters can be interpreted as the ability of the material to store (ϵ') and convert (ϵ'') the electric energy from the microwave irradiation.

The main mechanism of dielectric heating is attributed to the polarization losses that comprise dipolar, interfacial, ionic, and electronic losses [49,51]. All these contributions are related to the intrinsic properties of the materials. Thus, a corrected assessment of the microwave interaction with a particular material must include a detailed analysis of its dielectric properties. Additionally, if the material to be irradiated has high conductivity, then the conduction loss enters the summation in the dielectric loss constant (ϵ''). Magnetic contributions must be considered when the material presents a magnetic moment. Most luminescent materials are not intrinsically magnetic, making it possible to simplify the equation to only the electric contribution.

The interaction of microwaves with matter can be understood as how much a microwave can penetrate in a specific material. The penetration depth of a microwave within materials interacting only with the electric field ($\mu''_r = 0$) can be expressed by Equation (1):

$$dp = \frac{c}{\sqrt{2\pi f \sqrt{\epsilon'_r \mu'_r}}} \left[\left(\sqrt{1 + \tan^2 \delta_e} \right) - 1 \right]^{-1/2} \quad (1)$$

where f is the frequency of the microwave, (ϵ'_r) is the relative dielectric constant, and (ϵ''_r) is the relative dielectric loss. The loss tangent ($\tan \delta_e$) is an important parameter that denotes the dissipation of the electrical energy quantitatively due to multiple physical processes [58], and can be defined by Equation (2):

$$\tan \delta_e = \frac{\epsilon''_r}{\epsilon'_r} \quad (2)$$

The dp parameter will then classify the materials' interaction with the microwave in three different classes, (i) opaque or reflector, (ii) transparent, and (iii) absorber [49]. If the penetration depth is shallow (micrometer range), the microwave is reflected by the material's surface. Typical examples of microwave reflectors are metals like Cu ($dp = 1.3 \mu\text{m}$) and Al ($dp = 1.7 \mu\text{m}$). Alternatively, if the penetration depth is too large, then the microwave goes through the material without any interaction being transmitted. Materials like alumina, borosilicate glasses, and zirconia are excellent examples of microwave-transparent materials, with penetration depths larger than 10 m.

Neither of these situations is ideal for heating the materials and therefore performing microwave-assisted synthesis. To interact with microwaves, the radiation's penetration depth must be optimal to dissipate the electrical energy efficiently. Absorber materials have penetration depths in the order of centimeters, such as SiC ($dp = 1.9 \text{ cm}$), water (3.0 cm), and carbon black (5.7 cm) [59–61]. The tangent loss of the materials in terms of reflector, transparent and absorber have typical values of $\tan \delta_e < 0.01$ (transparent), >0.1 (absorbers), and $\gg 1$ (reflectors).

The fact that the dielectric properties of the materials can vary with temperature increases possibilities of interacting with materials by changing their reactional medium temperature. For instance, yttria-stabilized zirconia is microwave-transparent up to 200 °C ($\tan \delta_e = 0.001$; $dp = 5.68 \text{ m}$), suffering a transition to an absorber material at 600 °C ($\tan \delta_e = 0.15$; $dp = 0.039 \text{ m}$). The principle of intrinsic tangent loss modulated by the materials' temperature is beneficial for synthetic purposes. A synthetic precursor consisting of a mixture of low-lossy materials (transparent) can be in close contact with a high-lossy (absorber) material that will initially interact with the microwave, generating local heating. The heat transfer to the low-lossy material can inflict a change in the tangent loss, transitioning from transparent to absorber. This type of heating can be denominated as hybrid microwave heating, and it is the most-used method in microwave-assisted solid-state (MASS) synthesis [49,51,62,63].

1.3. Properties of Microwave Heating

Microwave-assisted heating can be characterized by a combination of unique properties offering faster and more efficient heating, often leading to more crystalline, homogeneous materials [64]. Firstly, the direct energy conversion into heat leads to a smaller temperature gradient throughout the precursors. In conventional resistive heating, the large temperature gradients make the reactions sluggish, taking a long time to reach equilibrium and generate the products. Dielectric heating works in the opposite manner, heating the materials from the inside to the outside. Dielectric heating leads to minimal energy consumption and faster reaction rates [65]. The thermal efficiency is an appealing point when considering laboratory work for sample screening and compositional tailoring. High throughput can be obtained with specialized setups, allowing the synthesis of tens of samples in few minutes. The fast screening in compositional tailoring can be beneficial for luminescent materials with a strong correlation of the optical property with the composition and structure.

Dielectric heating is also volumetric by nature, meaning that the heating is uniform throughout the sample, in theory. In actual samples, the volumetric heating is not perfect due to inhomogeneity in the precursors and temperature dependence of the dielectric properties [66]. However, compared with conventional heating, the MASS method has more uniform heating and, therefore, a more homogeneous composition of the final products.

For luminescent materials, this property is of utmost importance once the homogeneous dispersion of activator and sensitizer ions in the material's host leads to higher quantum yields and increased brightness [67]. Homogeneous luminescent materials offer a more-defined local structure that can lead to a precise control of the emitting color of crystal field-sensitive ions, such as Eu^{2+} [57].

The ionic diffusion limit encountered in conventional heating methods is also a significant problem in solid-state synthesis [64,68]. Usually, the diffusion is exceptionally slow if the temperature of the synthesis is lower than two thirds of the precursor's melting point (Tamann's rule) [69], which translates to temperatures usually higher than 1000 °C. The microwave radiation effect on ionic diffusion in bulk inorganic materials was investigated by Whittaker et al. [70]. The use of polarized MW irradiation combined with X-ray fluorescence technique has shown a preferential ionic migration in the polarized axis of the radiation (Figure 4a–b).

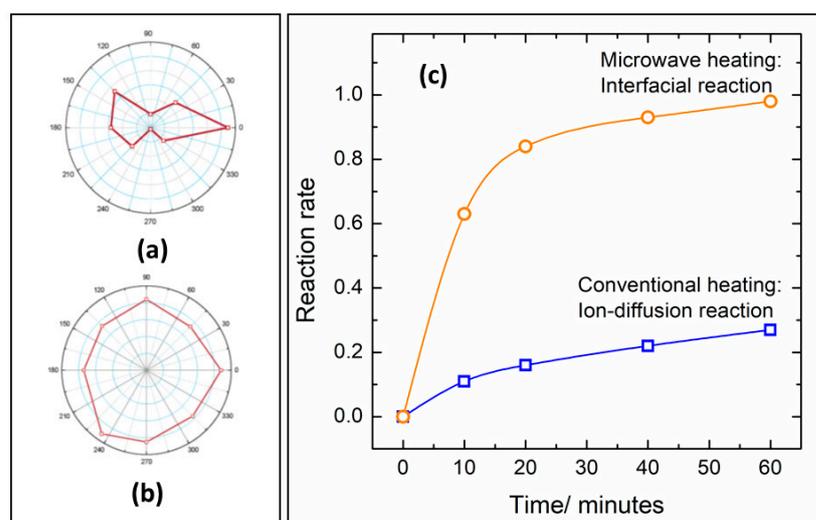


Figure 4. Ionic diffusion in a ceramic heated by polarized microwave irradiation (a) and conventional heating (b). Adapted from reference [70]. Reaction rates of NiFe_2O_4 synthesis under microwave heating (orange) and resistive heating (blue) at 850 °C (c). Adapted from ref. [71]. Copyright © 2005, American Chemical Society.

The increased diffusion is related to increased reaction rates in microwave-heated synthesis in comparison with resistive furnaces. Vanetsev et al. [71], studied the kinetics of NiFe_2O_4 formation by conventional and microwave-assisted solid-state reaction. Besides the faster reaction rates, microwave irradiation also promotes changes in the rate-controlling stage of the synthesis. By removing the diffusion hindrances due to non-thermal ponderomotive forces, the rate-controlling step of the synthesis changes from ion-diffusion to the interfacial chemical reaction step (Figure 4c) [71].

Instantaneous heating is also an essential feature of dielectric heating, swiftly converting the MW energy into heat. Correspondingly, once the microwave irradiation is ceased, the heating stops immediately, causing the reaction to quench instantaneously. This could lead to metastable phases, otherwise impossible to synthesize using conventional methods. The temperature quenching can be greatly beneficial for persistent luminescence phosphors because of the defect structure caused by rapid heating and cooling rates.

Once the dielectric constant and tangent loss vary with the material composition and structure, microwave irradiation allows for selective heating of the reactants [64]. This effect can find incredible applications on heating specific sites in supported metal catalysts [72]. The synthesis of chalcogenides greatly benefits the selective heating, allowing the reaction to occur before the sulfur evaporation [73].

The properties of microwave-assisted heating are summarized in Figure 5. The combination of features allows for the synthesis of highly efficient luminescent materials with singular properties, as will be discussed in detail in the next sections.

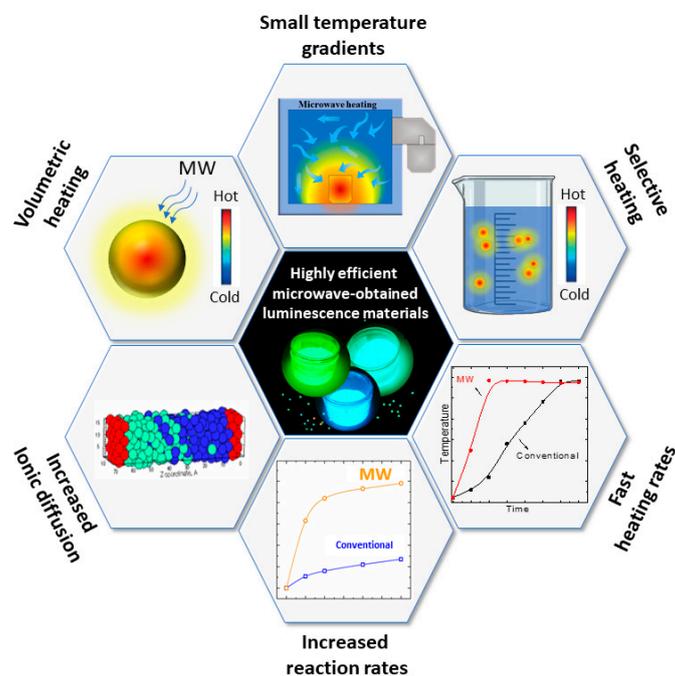


Figure 5. Fundamental microwave-assisted heating properties that are beneficial in preparing luminescent materials.

2. Methodologies of Microwave-Assisted Synthesis

There are several approaches to obtaining luminescent materials using microwave-assisted methodologies. However, microwave-assisted syntheses needs a detailed design of the experimental apparatus due to many dielectric heating parameters. The synthesis setup needs to be modified depending on the application to achieve optimum performance and reproducible results. Here, we present some details on the most common methods to obtain solid inorganic luminescent materials using microwave-assisted heating. Each one has advantages and disadvantages, and the materials' application should be considered when choosing the method as will be discussed further in next sections.

2.1. Microwave-Assisted Solid-State (MASS) Synthesis

Although there are some commercial setups for MASS synthesis, most of the literature relies on the usage of a domestic microwave oven (DMO) [49,54,55,63,74,75]. This is probably because of two main reasons: (i) economic viability: DMOs are inexpensive equipment, and (ii) DMOs are easily customizable to meet the needs of many methods.

The usage of DMOs comes with the price of several safety issues that need to be addressed carefully. In a customized DMO setup, the researcher needs to avoid radiation leakage, harmful to human body tissues at the frequency of 2.45 GHz. Additionally, DMO setups should be placed in a fume hood to avoid inhalation of toxic gasses from reactions. Effective thermal insulation of the reactional vessel is indispensable to prevent damage to the microwave oven.

To achieve optimum reproducibility, one needs to be consistent with the parameters of the synthesis, with a carefully chosen irradiation program. Crucial synthesis parameters are the crucible sizes, amount of starting materials, nature and mass of the susceptor, position inside the microwave, radiation frequency, and irradiation power levels [63].

A typical setup for microwave-assisted solid-state synthesis is depicted in Figure 6a,b. The setup for MASS (Figure 6b) synthesis consists of three parts [76,77]:

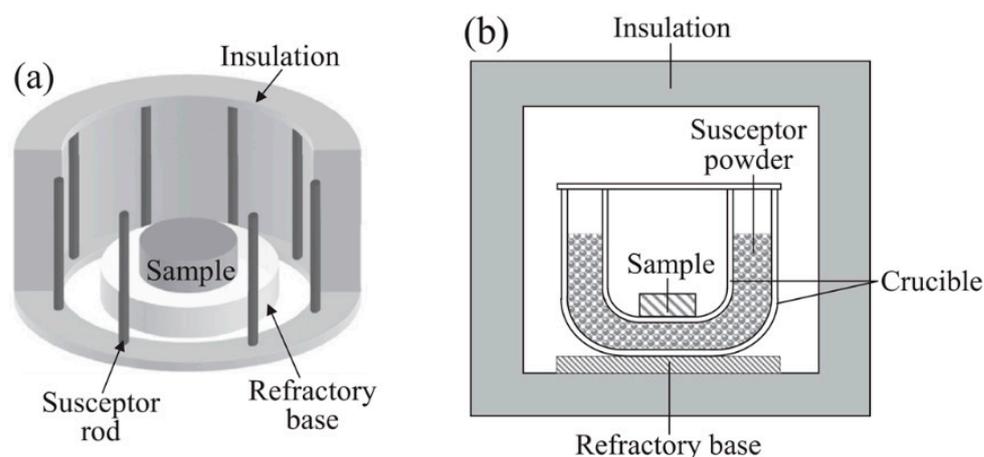


Figure 6. (a) Microwave-assisted solid-state synthesis setup with hybrid heating using SiC rods as susceptors. (b) MASS synthesis setup using a DMO. Susceptor surrounding the sample absorbs the microwave and initiates the heating [49]. Copyright © 2016, with permission from Elsevier.

(i) Reaction vessel: crucible where the precursor powder will be placed, and the reaction occurs. Usually, the crucibles are made of alumina (Al_2O_3) due to its high thermal and chemical stability. For specific purposes other crucibles can be used, such as zirconia or graphite. The ideal size of the crucibles is up for debate, but crucibles varying from 3 to 10 mL in volume are the most common.

(ii) Susceptor: material that absorbs 2.45 GHz microwaves efficiently at room temperature. Due to the fixed frequency of the radiation in DMOs, few oxides directly interact with the microwave. Since a significant number of luminescent materials use oxides as starting materials, most of the works in literature use the hybrid heating approach, where a susceptor (e.g., carbon) is used to initiate the heating (Table 1). The susceptor also has a secondary effect of generating a local atmosphere. The composition of the local atmosphere must be considered when synthesizing the materials. For example, a mildly reducing atmosphere is obtained when using carbon as a susceptor due to its incomplete burn that generates CO gas.

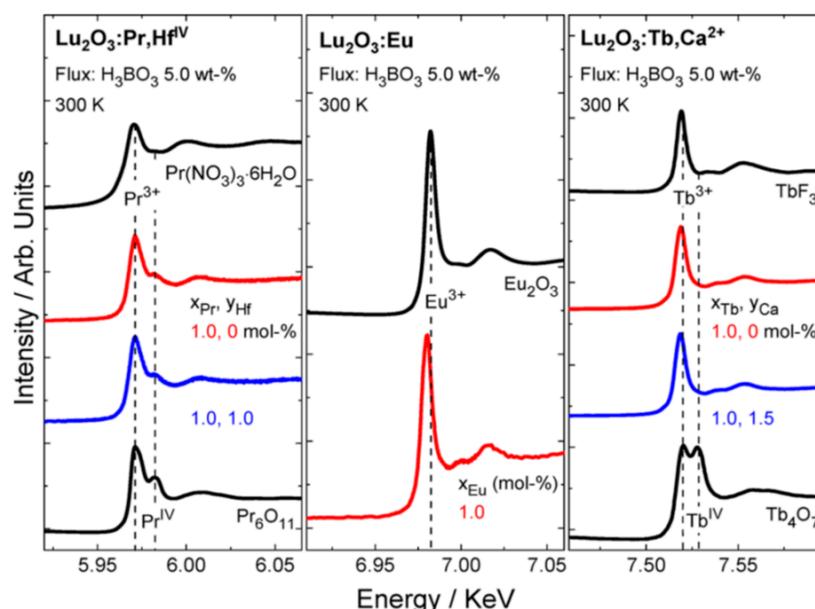
(iii) Thermal insulation: Insulating blocks where the reaction and susceptor crucibles will be placed to insulate extreme temperatures of the solid-state synthesis. Usually aluminosilicate bricks are used to host the reaction system (precursors + susceptors). Alumina foam and glass wool are also reported in the literature. The criteria to choose the correct system is the maximum thermal insulation allied to the high MW transparency.

The three-part system is inserted in a DMO and irradiated using time vs. power level program usually defined by measuring the temperature increase rate. Although temperature measurement is not an easy task, it is of utmost importance to avoid underreacted samples or decomposition by excessive temperatures [76]. A pyrometer is a convenient tool to measure the instantaneous temperature of the crucible. However, the recorded temperature is just of the material's surface. The temperature inside the susceptor should be considerably higher because the microwave penetration promotes heating initially below the surface of the material [78]. The same effect can be expected for the sample, where the core is hotter than the surface. Despite the difference in temperature between the core and surface, the use of hybrid heating using a susceptor helps obtain a more homogeneous heating of the samples compared to direct heating [78].

Table 1. Microwave dielectric properties ($\tan \delta_e$ and dp) and average heating rate of selected susceptor materials. The values are from the references [76,79–81].

Material	$\tan \delta_e$	dp (m)	Average Heating Rate ($^{\circ}\text{C/s}$)
Amorphous Carbon	-	-	21
Graphite powder	0.36–0.67	0.013–0.002	10
Activated Carbon	0.31–0.9	0.007–0.03	4.8
Fe_3O_4 (Magnetite)	0.02	-	7.6
CuO	0.087	0.21	2.7
SiC	0.388	0.019	0.3–2.8
Co_2O_3	0.0002	-	7.2
NiO	0.001	-	3.5

Carbon is the most-used susceptor and can absorb microwave radiation from DMOs reaching rapidly high temperatures (Table 1). Additionally, carbon undergoes through an incomplete burn at high temperatures, generating CO reducing gas. The local, reducing atmosphere can be beneficial if the luminescent material to be prepared needs to reduce activator ions' initial oxidation state. For instance, Eu^{2+} doped in $\text{Sr}_2\text{MgSi}_2\text{O}_7$ phosphors, which need to be reduced from the trivalent state in the precursor, e.g., Eu_2O_3 [57]. Tb_4O_7 is a common precursor for Tb-doped phosphors prepared by the solid-state method, sometimes needing H_2 gas to be reduced from Tb^{4+} to Tb^{3+} [82]. With the MASS method, it is possible to achieve $\text{Lu}_2\text{O}_3:\text{Tb}^{3+}$ in a pure trivalent state, in 20 min, using activated carbon as susceptor (Figure 7) [62]. When preparing $\text{Y}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$, the pure tetravalent form of cerium oxide, CeO_2 , is reduced to Ce^{3+} during the MASS synthesis in a single step of ca. 20 min [83].

**Figure 7.** L_{III} edge XANES spectra of Pr (left), Eu (middle), and Tb (right) doped Lu_2O_3 . Spectra for reference materials are also provided [62]. Copyright © 2016, American Chemical Society.

Silicon carbide, SiC, is also a good option for susceptors because of its high-temperature stability ($2830\text{ }^{\circ}\text{C}$). SiC is common in commercial microwave ashing furnaces that use SiC plates or rods to convert the MW radiation into heat (Figure 6a) [84]. Despite a slow initial heating rate, SiC has a monotonic increase of temperature, from 25 to $600\text{ }^{\circ}\text{C}$, working as a steady microwave heating element. After $600\text{ }^{\circ}\text{C}$, the microwave heating rate increases dramatically, reaching $1800\text{ }^{\circ}\text{C}$ in few minutes [80]. CuO has some usage as a reactant-susceptor, used as a reactant in the MASS synthesis of $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ superconductors [53].

2.2. Microwave-Assisted Combustion (MAC)

Combustion relies on rapid oxidation that generates heat or as a sequence of exothermic chemical reactions between a fuel and an oxidant that is accompanied by the production of heat and the conversion of chemical species [85–88]. In the volume reaction model, also referred to as thermal explosion, the entire sample is uniformly heated to the ignition temperature, and the combustion reaction coincides in all parts of the sample.

The conventional combustion synthesis uses a certain amount of oxidizer (e.g., metal nitrates) and fuel (e.g., urea) to produce an aqueous solution that, upon heat, ignites the combustion [87,88]. The microwave-assisted method works in the same way, but the formed solution needs to absorb the microwave radiation to initiate the heat. Several solvents (Table 2) have a suitable dielectric tangent loss to interact with the microwave electric field and convert it into heat.

Table 2. Dielectric tangent loss ($\tan \delta_e$) of various solvents. Values were obtained at 20 °C and 2.45 GHz microwave frequency [56].

Solvents	$\tan \delta_e$
Water	0.15
Ethylene glycol	1.35
Ethanol	0.941
Methanol	0.659
DMSO	0.825
1-Butanol	0.571
DMF	0.161
Acetic Acid	0.174
Dichloromethane	0.042

The combustion synthesis has many advantages such as high reaction temperatures with relatively low preheating temperatures, high heating rates, short duration of reactions, relatively simple equipment, materials with porous morphology, and varied microstructure [89]. The microwave-assisted combustion makes the synthesis even faster, in the order of tenths of seconds (Figure 8).

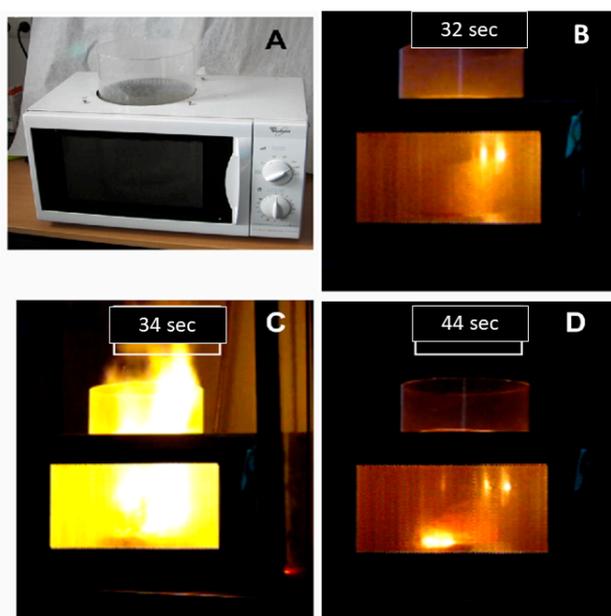


Figure 8. Microwave-assisted combustion synthesis setup using DMO equipment (A). Synthesis progression at 32 (B), 34 (C), and 44 s (D). Such extensive modification in DMOs needs thorough consideration of the involved hazards, mainly microwave leakage. Reproduced from ref. [85,90] with permission from The Royal Society of Chemistry.

The experimental setup for the MAC method can be obtained with a modified DMO. Nevertheless, attention to the safety issues must be thoroughly considered due to physical and chemical hazards involved in the combustion flames and toxic fumes. Moreover, microwave leakage from the modified setup should be avoided at all costs. An example of the synthesis configuration used for microwave-assisted combustion synthesis is depicted in Figure 8.

2.3. Microwave-Assisted Hydrothermal (MAH)

Hydrothermal synthesis refers to a crystallization technique of compounds under high-temperature aqueous solutions and at high vapor pressures. The crystal growth is performed in an apparatus consisting of a steel pressure vessel called an autoclave. A temperature gradient is maintained between the opposite ends of the chamber where the hotter end dissolves the ions. At the cooler end the crystals are formed. Advantages of the hydrothermal method include the ability to create crystalline phases which are not stable at the water boiling point.

The vessel barrier in the hydrothermal apparatus makes the synthesis in a resistive oven long and energetically inefficient. On the other hand, the microwave-assisted hydrothermal (MAH) method is an advantageous and powerful method to synthesize inorganic materials in a noticeably short time. The microwave radiation interacts directly with the water heating the solution in a short time. It makes the MAH synthesis highly efficient regarding energy conversion. The pressured solution allied to high temperatures allows crystallization of particles in the nanoscale as well. The vast number of compounds soluble in water, temperature modulation and pressure control allow researchers obtain nanoparticles with different composition, crystalline structures and morphologic shapes (rings, crystals, rods, spheres, cubes, urchin) [91].

Although there are examples of MAH synthesis using a modified DMO, it is advised to use a specialized setup due to the risk of explosion of the pressurized vessel [92]. Luckily, high-quality commercial setups are available. Among the advantages of MAH synthesis, the most surprising is obtaining nanoparticles with a narrow particle size distribution and varied morphologies (Figure 9). This is possible due to the homogeneous and fast dielectric heating by the microwave radiation.

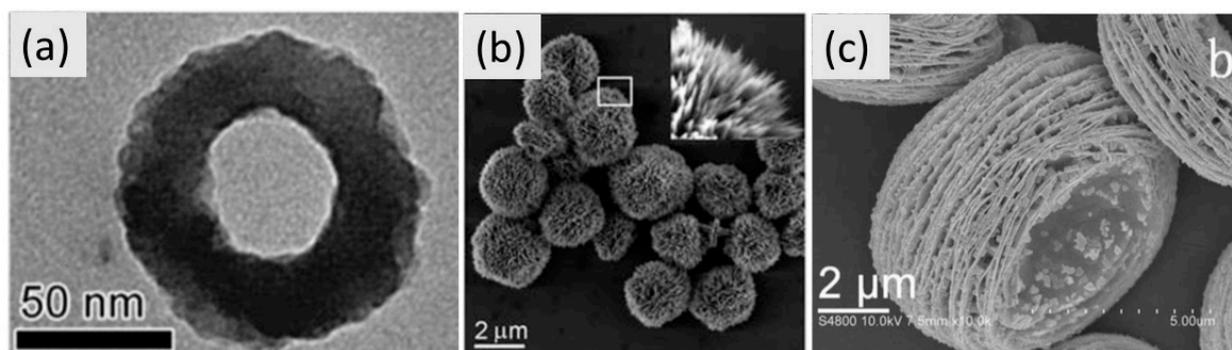


Figure 9. α - Fe_2O_3 nanorings (a), CuO nanourchins (b), and lamellar $\text{Cd}(\text{OH})_2$ (c) obtained by microwave-assisted hydrothermal method. Adapted from ref. [91]. Copyright © 2016, with permission from Elsevier.

2.4. Microwave-Assisted Sol-Gel (MASG)

Sol-gel synthesis combines two synthetical methodologies. In the first step, a polymeric 3D-framework is obtained by using organometallics or silane precursors. These precursors undergo hydrolysis and condensation process to form a gel, that contains evenly dispersed ions in the polymeric matrix. In a second step, to obtain the final material the gel precursor is thermally treated to remove carbon moieties. In the conventional sol-gel method, the heat treatment is performed using a resistive furnace. The MASG method utilizes microwave-assisted heating to thermally treat the gel precursors.

Birkel et al. [93]. obtained Pd-substituted LnFeO_3 (Ln: Y, and La) using the MASG method. The prepared gel using citric acid and metal acetylacetonates was dried and then ground to obtain a powder that was subsequently heated in a microwave using the MASS setup. The noticeably short heating times in microwave reactions allowed the elucidation of the phase formation process showing that the reaction for forming LaFeO_3 is completed after 90 s. In comparison with the furnace-based sol-gel or combustion methods, the MASG method has allowed the easy and reproducible tuning of material properties such as crystallite size and surface area, which are important parameters for photocatalytic activity.

Near-infrared (NIR) persistent luminescence nanoparticles are widely investigated for bioimaging and theranostics. However, if the particle's hydrodynamic radius is bigger than 100 nm, the particles are sequestered by the reticuloendothelial system (RES). Non-aqueous MASG synthesis can be an efficient route to obtain ultra-small nanoparticles (average size: 6 nm) of $\text{ZnGa}_2\text{O}_4:\text{Cr}^{3+}$ persistent phosphors [94]. The ultra-small nanoparticle size reduces the RES clearance, enhances the blood circulation time, and allows widespread organ distribution. The non-aqueous benzyl alcohol route has been established as a powerful technique that takes place at a moderate temperature and pressure to obtain exceedingly small metal oxide nanoparticles with high crystallinity, purity, and reproducibility. The combination with the microwave heating offers a reduction of the reaction time (30 min) in comparison to traditional heating in an autoclave (48 h) [95]. One special advantage of this technique is that the one-pot process does not need further high temperature treatment, which generally leads to excessive particle growing.

2.5. Other Preparation Methods

Coupled methodologies with microwave-assisted heating are an interesting alternative for synthesizing luminescent materials. Bi et al. [96] used a coupled microwave-ultrasound-assisted method to obtain luminescent $\text{Y}_2\text{O}_3:\text{Eu}^{3+}$ materials. The method consists of the irradiation of a $\text{Y}(\text{NO}_3)_3$ water/ethanol solution using ultrasound (100 W) and microwave (150 W) for one hour. The method allows for nanoparticle formation (<100 nm) due to the local heating leading to fast crystallization and the ultrasound avoiding the excessive coalition of smaller particles. In this case, the obtained luminescent materials present high quantum yields of emission (ca. 80%).

The co-precipitation method consists of precipitate substances normally soluble in the precursor solution. In this method the mixing of the solution containing the soluble cations and anions forces the parts to interact, inducing nucleation and growth. After the particles surpass the critical mass the precipitation of the materials is observed. Usually aqueous solutions are used; however, it is possible to obtain precipitates in other polar solvents. Thongtem et al. [97] used a microwave-assisted co-precipitation method to obtain SrMO_4 (M: W, and Mo) nanocrystals. The crystallization of the materials was performed by irradiating an ethylene glycol (EG) dispersion of the NaMO_4 and $\text{Sr}(\text{NO}_3)_2$ salts. The effect of the EG solvent in the synthesis is not discussed, but the mildly chelating effect might play a role in retarding the precipitation and crystallization. Moreover, EG is an excellent microwave absorber, leading to a homogeneous heating profile. Due to the controlled crystallization in this method, they were able to produce small nanoparticles (~25 nm) with a narrow size distribution.

The comparison methods of main microwave-assisted methods is summarized in Table 3.

Table 3. Comparison of microwave-assisted methodologies used to obtain luminescent inorganic materials.

Method *	Advantages	Disadvantages	Benefits for Luminescent Materials	Examples
MASS	<ul style="list-style-type: none"> Energy saving Fast processing times (minutes) Increased reaction rates Increased ionic diffusion High homogeneity 	<ul style="list-style-type: none"> Few commercial setups available DMOs can struggle with the reproducibility In modified DMOs safety issues must be addressed 	<ul style="list-style-type: none"> High QY High storage capacity Increased lifetimes Less prone to concentration quenching 	<ul style="list-style-type: none"> $Y_3Al_5O_{12}:Ce^{3+}$ [83] $Sr_2MgSi_2O_7:Eu^{2+}, Dy^{3+}$ [57]
MAC	<ul style="list-style-type: none"> Fast processing times (seconds) Porous materials Reduced particle sizes 	<ul style="list-style-type: none"> Morphology control is difficult Carbon residues In modified DMOs safety issues must be carefully addressed 	<ul style="list-style-type: none"> Combination of high homogeneity and moderately small particle sizes 	<ul style="list-style-type: none"> $Y_3Al_5O_{12}:Er^{3+}, Yb^{3+}$ [85] $Gd_2O_3:Er^{3+}, Yb^{3+}, Zn^{2+}$ [90]
MASG	<ul style="list-style-type: none"> Fast processing times (minutes) Nanoparticulate materials Morphology control 	<ul style="list-style-type: none"> Carbon residues Expensive and reactive precursors 	<ul style="list-style-type: none"> Nanoparticulate materials Possible to use as a coating method 	<ul style="list-style-type: none"> $ZnGa_2O_4:Cr^{3+}$ [94] $M_2SiO_4:Eu^{2+}$ (M:Ca,Ba) [98]
MAH	<ul style="list-style-type: none"> Fast processing times (Seconds to minutes) Nanoparticulate materials Finely controlled morphology 	<ul style="list-style-type: none"> DMOs cannot be used due to safety issues Expensive commercial setups 	<ul style="list-style-type: none"> Nanoparticles for increased applicability in bioimaging and photodynamic therapy 	<ul style="list-style-type: none"> $YVO_4:Eu^{3+}$ [99] $ZnWO_4:Dy^{3+}$ [100]

* Microwave-assisted solid-state (MASS), combustion (MAC), sol-gel (MASG), and hydrothermal (MAH).

3. Inorganic Luminescent Materials Obtained by Microwave-Assisted Methods

Inorganic luminescent materials can be obtained by various methods, such as conventional solid-state (CSS) synthesis, combustion synthesis, or sol-gel methods. However, the CSS method is the most widely used due to the combination of the easy processing and few steps that grants high reproducibility. In the CSS method, a high-temperature, direct reaction is expected where the solid reactants, with different grain sizes and randomly oriented surfaces, must react through ionic diffusion. The drawback of the solid-state reactions is the limited flux of diffusing species of precursors, impeding crystal growth and forming the final products [57]. The combination of low ionic diffusion with large temperature gradient during the resistive heating translates to long synthesis' time at high temperatures (>1000 °C) to avoid inhomogeneous materials with poor optical properties. Thus, increasing the ionic diffusion and reducing the temperature gradient can be an alternative method of obtaining efficient inorganic luminescent materials swiftly. In the following sections, we review several classes of luminescent materials obtained by microwave-assisted methodologies.

3.1. Oxides

Several luminescent materials are based on oxides doped with activator ions, e.g., $Y_2O_3:Eu^{3+}$ [96], $Lu_2O_3:Tb^{3+}$ [101]. Usually, oxide luminescent materials are prepared from oxide precursors in a solid-state synthesis. In addition, carbonate precursors are commonly used with the same intent to obtain the respective oxide after the decomposition of carbon moieties at moderate temperatures. Since most of the metal oxides are low-lossy materials, having considerable penetration depth, they act as microwave-transparent materials. Thus, hybrid heating is used to achieve the needed high temperature for the synthesis.

Pedroso et al. [62] used the MASS method to obtain the $Lu_2O_3:RE^{3+}$ (RE: Pr, Eu, and Tb) persistent luminescent materials using a modified DMO setup. Rare-earth doped

lutetia is a known family of materials with long persistent luminescence decay, especially the $\text{Lu}_2\text{O}_3:\text{Tb}^{3+}$, Ca^{2+} [82]. Lu_2O_3 -based materials require rather extreme preparation conditions in solid-state synthesis needing high temperature ($\sim 1700^\circ\text{C}$) in high vacuum or $\text{H}_2\text{-N}_2$ gas flow [102]. When the MASS method is employed, the material can be prepared in one step of 22 min. Despite the short synthesis time, the materials are highly crystalline and have large storage capacity compared to the materials prepared by the conventional solid-state method (Figure 10). Most importantly, complete reduction of Tb_4O_7 to Tb^{3+} was achieved by the reducing atmosphere generated in-situ during the synthesis.

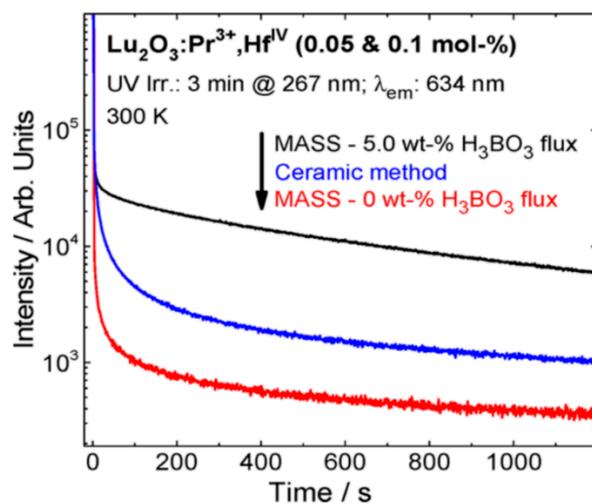


Figure 10. Persistent luminescence decay curves of $\text{Lu}_2\text{O}_3:\text{Pr}^{3+}$, Hf^{4+} materials obtained by conventional (blue curve), and microwave-assisted (black curve) solid-state methods, and the PeL decay of the MASS obtained materials without H_3BO_3 flux (red curve) [62]. Copyright © 2016, American Chemical Society.

3.2. Aluminates

Aluminates are commonly used as a host for LED and energy storage phosphors. The benchmark persistent luminescence material is a doped strontium aluminate with the general formula $\text{SrAl}_2\text{O}_4:\text{Eu}^{2+}$, Dy^{3+} [103] that has the best efficiency to date. Interestingly, the canonical material for pc-LEDs is also an aluminate, $\text{Y}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$ [104]. In 2017, Finley et al. [105] used microwave-assisted heating to synthesize $\text{SrAl}_2\text{O}_4:\text{Eu}^{2+}$, Dy^{3+} phosphors starting from reverse micelle precursor. The reverse micelle was prepared in an aqueous metal nitrates solution with a combination of components for emulsion (CTAB, n-Heptane, and 1-butanol). The prepared reverse micelle was submitted to a hybrid microwave heating using activated carbon as a susceptor. The total irradiation time was 14 min, yielding nearly phase-pure materials. The combined reverse micelle with microwave-assisted heating produced particle sizes 70% smaller ($d_{0.5} = 4.2 \mu\text{m}$) than the conventional MASS ($d_{0.5} = 14.3 \mu\text{m}$). Despite the similar photoluminescence properties, the reverse micelle material has a higher temperature quenching and slightly larger storage capacity.

In 2012, Birkel et al. [83] used the MASS method to prepare $\text{RE}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$ (RE: Y, and Lu) starting from RE_2O_3 and Al_2O_3 , and small quantities of fluoride fluxes. The synthesis is efficient for obtaining highly crystalline garnet phosphors with similar quantum yields to conventional solid-state samples. Usually, luminescent rare earth aluminum garnets need thermal treatment at high temperatures (1500°C) for several hours to form the material. On the other hand, the MASS synthesis provided an energy saving of 99% with nearly identical optical and structural quality (Figure 11). The quantum yields for the MASS-obtained aluminum garnets were 88% and particle sizes were slightly smaller.

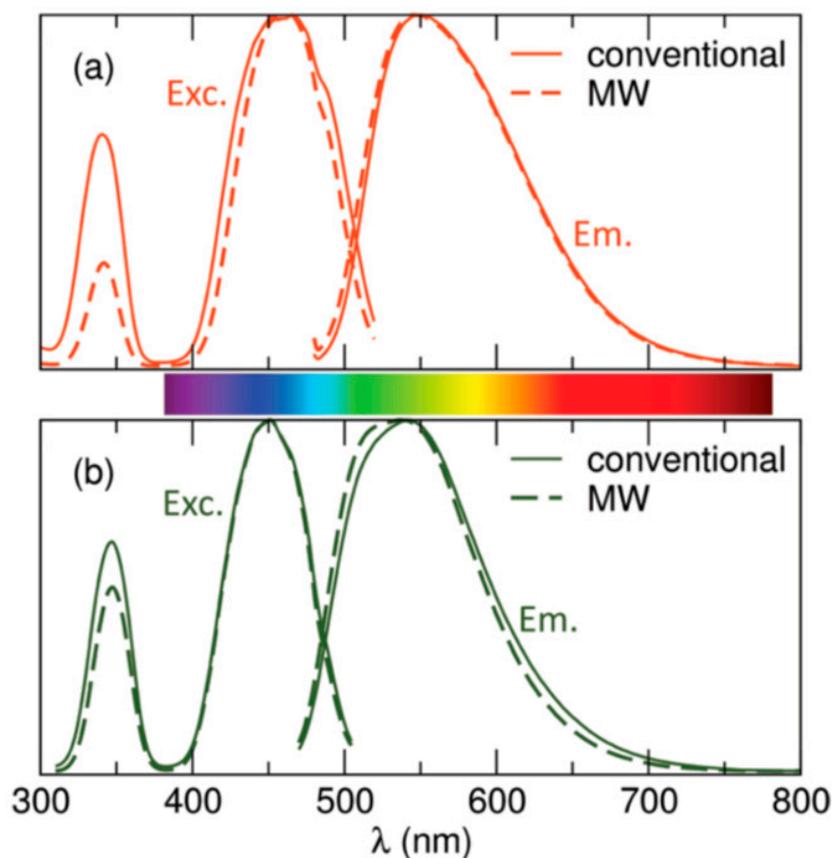


Figure 11. Excitation and emission spectra of $\text{Y}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$ (a) and $\text{Lu}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$ (b) obtained by both conventional (line) and MASS (dashed line) methods [83]. Copyright © 2012, American Chemical Society.

3.3. Silicates

Silicate is an essential class of inorganic host matrices for luminescent materials. The most famous example is the efficient blue-emitting $\text{Sr}_2\text{MgSi}_2\text{O}_7:\text{Eu}^{2+}$ phosphor.

In 2020, Carvalho et al. [57] studied the MASS synthesis of $\text{Sr}_2\text{MgSi}_2\text{O}_7:\text{Eu}^{2+}$, Dy^{3+} persistent phosphors and microwave irradiation's effects on the homogeneity and storage capacity. Comparing the conventional and microwave-assisted methods on obtaining the $\text{Sr}_2\text{MgSi}_2\text{O}_7:\text{Eu}^{2+}$, Dy^{3+} shows that the MASS method provides higher dopant homogeneity in the crystal due to increased ionic diffusion induced by microwave irradiation. However, the local atmosphere generated by carbon susceptor is sometimes inefficient to reduce the activator ions. In this case, decoupled, two-step synthesis is beneficial to obtain $\text{Sr}_2\text{MgSi}_2\text{O}_7:\text{Eu}^{2+}$, Dy^{3+} materials with high quantum yields, and large storage capacities. In the first step, highly homogeneous materials are obtained due to the increased diffusion in MASS synthesis (Figure 12). The second step reduces the activator ions in mild temperatures achieving optimum Eu^{2+} concentrations for high-brightness and a large storage capacity.

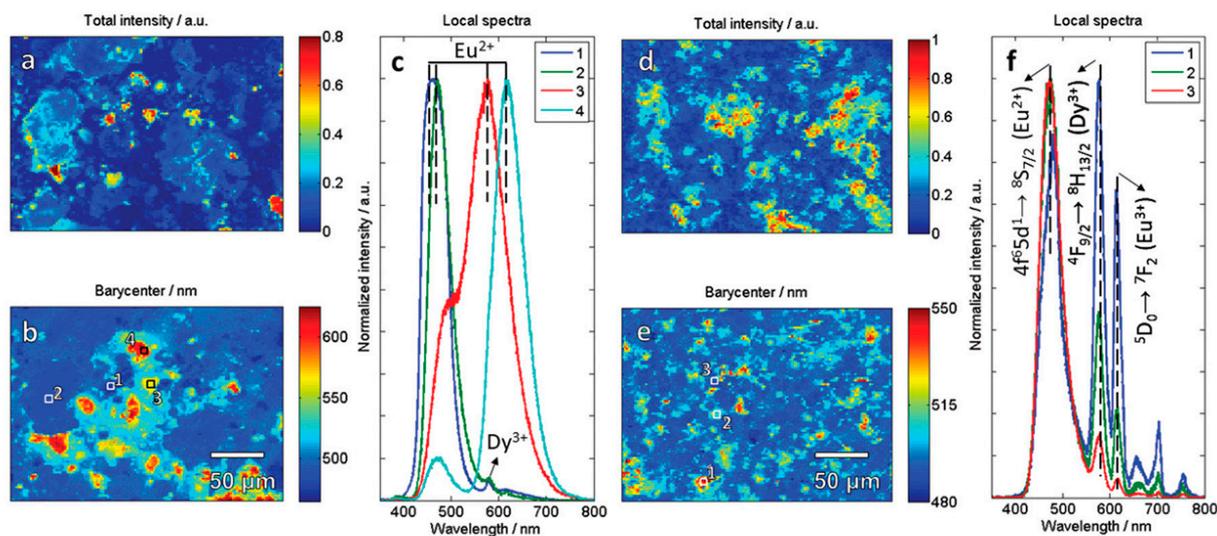


Figure 12. SEM-CL hyperspectral images of the $\text{Sr}_2\text{MgSi}_2\text{O}_7:\text{Eu}^{2+}(1\%), \text{Dy}^{3+}(1\%)$ materials obtained by CSS (a,b) and MASS (d,e) method. The numbers in the images (b,e) correspond to the area of the local spectra extracted for the CSS (c) and MASS (f) materials. Reproduced from Ref. [57] with permission from The Royal Society of Chemistry.

The MASS method can also be beneficial when the synthesis's precursor is volatile or decomposes at high temperatures. The swift nature of the dielectric heating allows the reaction to occur before evaporation or degradation takes place. Brgoch et al. [106] obtained Ce^{3+} -doped silicates with complex composition, $\text{Na}_3\text{YSi}_2\text{O}_7$, using a MASS methodology. The rapid MASS method allows for synthesizing the materials using stoichiometric amounts of the precursors, unlike conventional preparations that must account for sodium carbonate's evaporation.

3.4. Aluminosilicate

Aluminosilicate consists in AlO_4 and SiO_4 units interconnected in a 3D framework. Famous example are the zeolites and minerals derived from the sodalite structure. Hackmanite, a sulfide containing sodalite, is a material that exhibits persistent luminescence when doped with Ti ions [107,108]. The white emission has long durations and is the record-long for non-rare earth persistent phosphors [109,110]. The complex structure is usually synthesized using a two-step solid-state synthesis that takes ca. 50 h combined. In 2019, the series of sodalite $(\text{Na},\text{M})_8\text{Al}_6\text{Si}_6\text{O}_{24}(\text{Cl},\text{S})_2$ (M: Li, K, Rb) materials were synthesized using a zeolite precursor applying the MASS method [111]. The structural similarities of the precursors and products, together with the enhanced ionic diffusion induced by MW radiation, reduce the synthesis' time to 18-20 min. The formation of the material goes through a crystalline intermediate, NaAlSiO_4 , showing a sharp transition from zeolite to sodalite after 8 min of MW irradiation.

Brgoch et al. studied the MASS synthesis of the Eu-doped $(\text{Ba}_{1-x}\text{Sr}_x)\text{Al}_2\text{Si}_2\text{O}_8$ aluminosilicate materials. The method yields nearly phase-pure materials, which are often difficult to achieve using conventional solid-state methods. Both of the phases, hexagonal and monoclinic, can be synthesized with this method by varying the irradiation time and power levels. Short irradiation (2 min) is enough to synthesize the hexagonal phase, whereas longer times (100 min) lead to the monoclinic phase (Figure 13). The order of the crystallization was interpreted using the Ostwald law wherein the less stable polymorph crystallizes first.

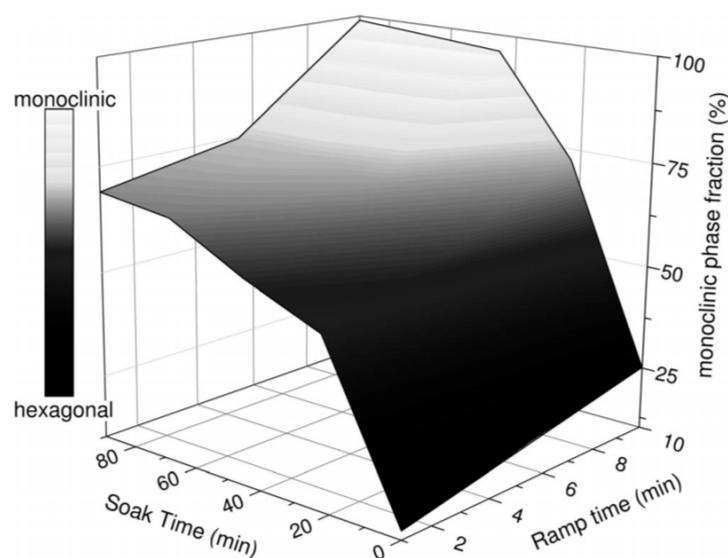


Figure 13. Contour plot of the conversion from the hexagonal phase to the monoclinic phase of Eu-doped $(\text{Ba}_{1-x}\text{Sr}_x)\text{Al}_2\text{Si}_2\text{O}_8$ materials. The ramp time and soak time (in minutes) of the microwave reaction are presented on the x and y axes, respectively, while the refined phase fractions are presented on the z axis [112]. Copyright © 2014, with permission from John Wiley and Sons.

3.5. Sulfides and Oxysulfides

Sulfides and oxysulfides inorganic materials are commonly used as host matrices for luminescent materials. The presence of sulfur, which is bigger and more polarizable compared to oxygen, makes these hosts interesting for emission color tuning of crystal field-sensitive activators, such as Eu^{2+} and Ce^{3+} .

The synthesis of alkaline earth sulfides is exceedingly complex, usually needing special synthesis setups. For instance, a reducing atmosphere composed by toxic H_2S gas is often needed. The usage of poisonous gas limits the synthesis to specialized furnaces. Moreover, the synthesis of the sulfides by conventional method must consider the evaporation of the sulfur during the process. In 2020, Santos et al. [73] synthesized SrS materials doped with various rare-earth ions to compare the co-dopant effect in PeL storage capacity. The MASS synthesis was applied successfully to obtain SrS materials in a single step without using H_2S to reduce the sulfur to sulfide anions. Although there were small SrSO_4 impurities, crystalline rare-earth-doped SrS materials were obtained with the facile method in 22 min. The decay time of the persistent luminescence was measured in luminance mode, indicating overall good performance of the PeL phosphors.

Sulfur-containing materials are prone to oxidation depending on the synthesis conditions. For instance, $\text{La}_2\text{O}_2\text{S}$ materials suffer oxidation to La_2OSO_4 at high temperatures in an oxygen-containing atmosphere [113]. Thus, the synthesis of sulfur-containing materials using MASS synthesis must be finely controlled to avoid excessive temperature rise. In 2018, Carvalho et al. [76] used the MASS method to obtain the $\text{RE}_2\text{O}_2\text{S}$ (RE: La, Gd, and Y) materials doped with $\text{Ti}^{3+/4+}$ and Mg^{2+} ions. After the 25 min irradiation, the crystalline phase is mainly the $\text{RE}_2\text{O}_2\text{S}$ hexagonal phase. The analysis of the sulfur K-edge XANES and FTIR spectra showed rapid oxidation of the elemental sulfur (S^0) to sulfate (S^{6+}) in the first 2 min. It is shown that the reduction takes place from 2 to 10 min, where only S^{2-} species are observed (Figure 14). The increased irradiation time promotes partial oxidation of S^{2-} to SO_4^{2-} species. The results showed a great versatility of the MASS synthesis of sulfur-containing materials where different irradiation times allied with the rapid heating and cooling rates can lead to different sulfur valences.

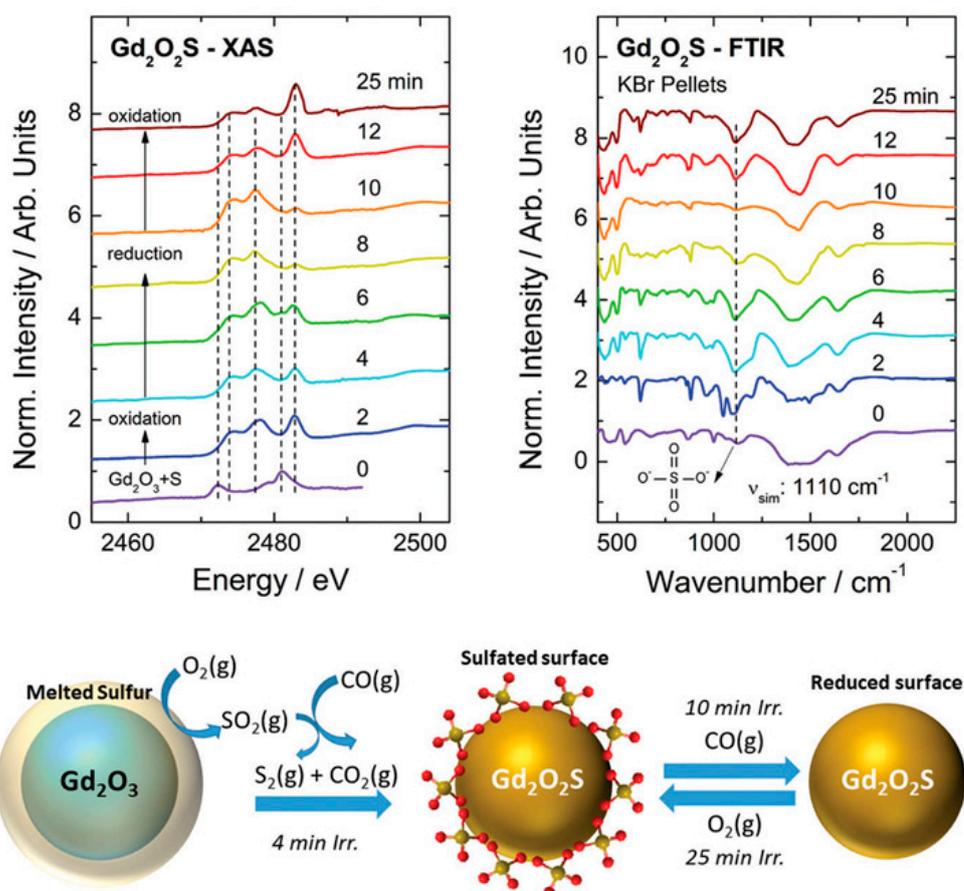


Figure 14. SR-XANES of the sulfur K-edge (**top left**) and FTIR spectra (**top right**) of the $\text{Gd}_2\text{O}_2\text{S}$ materials obtained by the MASS method for different durations of microwave exposure. Schematic representation of oxysulfide formation with the participation of the gases assisting the redox process of sulfur (**bottom**). Reproduced from Ref. [76] with permission from The Royal Society of Chemistry.

3.6. Oxometalates

Metalates are the name given to complex anions composed of a metal bonded to a different type of anions. In the case of the oxometallate, a transition metal is bonded to oxygen atoms to form complex anions in various forms, e.g., MnO_4^- , VO_4^{3-} , CrO_4^{2-} , WO_4^{2-} , and MoO_4^{2-} .

Rare earth-doped tungstate and molybdate matrices are extensively studied as luminescent materials, and are usually obtained by solution-phase methods, such as coprecipitation, sol-gel, hydrothermal, and Pechini. These methods generally require more than one step to separate the material from the solution or heat treating the materials to achieve phase-pure phosphors. Several scheelites have been prepared by using the microwave-assisted solution-phase methods, such as $\text{NaLa}(\text{MoO}_4)_2$ [114], $\text{Na}(\text{Ca},\text{Sr})(\text{La},\text{Gd})(\text{MoO}_4)_3$ [115,116], and $(\text{Sr},\text{Pb})(\text{La},\text{Gd},\text{Y})_2(\text{MoO}_4)_4$ [117]. Solid-state synthesis using microwave heating of oxometallates is scarce, with BaMoO_4 [118] and CdWO_4 [119] being some examples.

In 2018, Zhai et al. [120] prepared the $\text{CaMoO}_4:\text{Eu}^{3+}$, Dy^{3+} scheelite using the MASS method with carbon as susceptor and metal carbonates as precursors. The irradiation program was 30 min at 560 W power level, using a DMO. The synthesized materials possess interesting morphology where micrometric cubes are formed by close-packed quasi-spherical grains.

In 2020, Perera et al. [121] described the MASS synthesis of a series of $\text{NaRE}(\text{MO}_4)_2$ materials (RE: La, Pr, Eu, Dy; M: Mo, W). The synthesis was conducted using pure oxide precursors using various irradiation programs. The total time of the synthesis varied in

the interval of 6 to 27 min. The authors reported that the synthesis was robust, fast, and that it was simple to obtain structurally complex metallates. As for the luminescence properties, the materials obtained by MASS synthesis were compared to those from the conventional heating method, and the light output was nearly identical (Figure 15), apart from $\text{NaLa}_{0.95}\text{Eu}_{0.05}(\text{MoO}_4)_2$ material that exhibits higher emission intensity for MASS-obtained material.

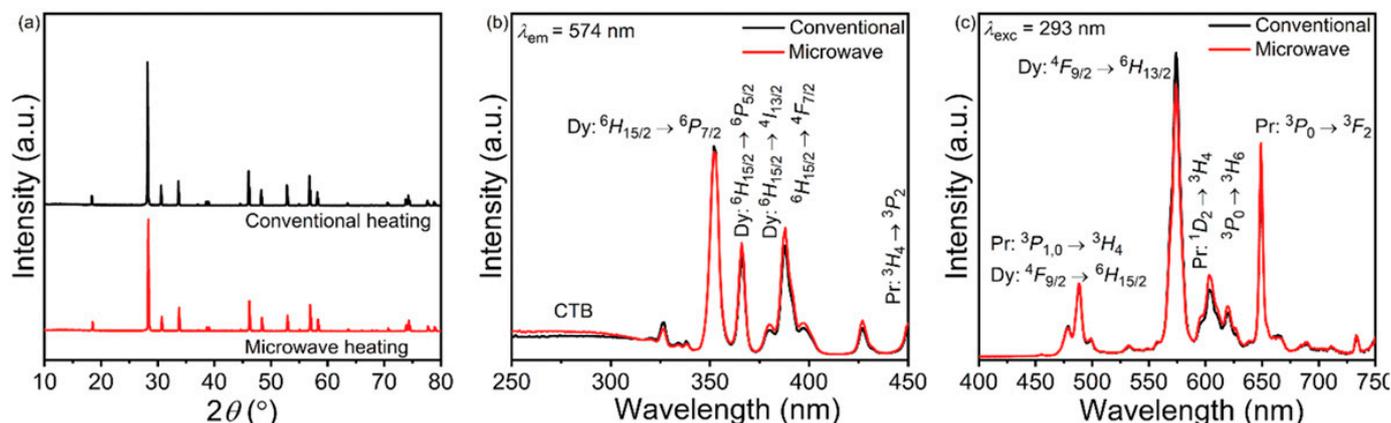


Figure 15. (a) XRD patterns, (b) excitation spectra, and (c) emission spectra of quinary molybdate phosphors $\text{NaLa}_{0.95}\text{Pr}_{0.025}\text{Dy}_{0.025}(\text{MoO}_4)_2$ synthesized using conventional and microwave heating. Reproduced from ref. [121] with permission from The Royal Society of Chemistry.

In 2015, Dué et al. used the microwave-assisted hydrothermal method to obtain luminescent nanovanadates with the formula $\text{Y}_{1-x}\text{Eu}_x\text{VO}_4$. The method yields small nanoparticles (~10–15 nm) in only 15 min of microwave irradiation. The materials were reported as having efficiencies higher than the state-of-the-art inorganic optical material in liquid phase detection of H_2O_2 .

3.7. Other Materials

Du et al. [122] used the MASS method to obtain a novel Mn^{4+} -doped double-perovskite germanate structure with the formula $\text{La}_2\text{MgGeO}_6$. The approach consisted of two steps, first irradiating the sample with microwaves for 40 min, and then conventional thermal treatment for 3 h. Mn^{4+} phosphors are emerging as a new class of efficient NIR emitters for bioimaging and night-vision surveillance. However, these materials' preparation can be complex as the manganese ions can assume different valence states. This approach was used because the hybrid heating using charcoal generates CO gas, which acts as a reductant of Mn^{4+} to Mn^{2+} , at least partly. The reduction is undesirable because it changes the optical performance and even kills the emission ultimately. The second step of mild temperature heating allows the oxidation of the Mn ions to the tetravalent form. Combining these two effects, it is possible to obtain a highly dispersed dopant, homogeneous crystal structure, and optimized luminescence performance.

Several classes of materials were prepared by microwave-assisted methodologies, and the selected examples are presented in Table 4. The description of the methods, microwave program, and susceptor can be used as a guide to plan a synthetic route using microwave-assisted methodologies.

Table 4. Inorganic hosts for luminescent materials obtained by microwave-assisted methodologies.

Class	Composition	Method	Susceptor	MW Program	Ref.
Aluminate	BaMgAl ₁₀ O ₁₇ :Eu ²⁺	MAC	Urea and Water	15 min @ 850 W	[123]
	Ca ₃ Al ₂ O ₆ :Eu ³⁺	MAC	Solution	Until Ignition @ 560 W	[124]
	Lu ₃ Al ₅ O ₁₂ :Ce ³⁺	MASS	Carbon	18–25 min@850–1300 W	[83]
	MgAl ₂ O ₄	MASS	Carbon	100 min@700 W	[75]
				9 min@960 W +	
	SrAl ₂ O ₄ :Eu ²⁺ , Dy ³⁺	MASS	Carbon	5 min @480 W	[105]
	Sr ₃ Al ₂ O ₆ :Eu ²⁺ , Dy ³⁺	MASS	Carbon	20 min @ 650 °C	[125]
	Y ₃ Al ₅ O ₁₂ :Ce ³⁺	MASS	Carbon	18–25 min@850–1300 W	[83]
	Y ₃ Al ₅ O ₁₂ :Ce ³⁺	MAC	Solution	5,10,15 min@900 W	[126]
	Y ₃ Al ₅ O ₁₂ :Er ³⁺ , Yb ³⁺	MAC	Solution	Not provided	[85]
ZnAl ₂ O ₄ :Cu ⁺	MAC	Solution	1 min@900 W	[127]	
Aluminosilicate	Al ₆ Si ₂ O ₁₃ :Eu ³⁺	MASG	Solution	30–45min@680–800 W	[128]
				10 min@1000 W +	
	(Ba _{1-x} Sr _x) _{1-y} Eu _y Al ₂ Si ₂ O ₈	MASS	Carbon	30–90 min@375 W	[112]
	(Na,M) ₈ Al ₆ Si ₆ O ₂₄ (Cl,S) ₂	MASS	Carbon	18 min@450 W	[111]
Fluoride	KMF ₃ (M = Zn, Mg, Mn, Co)	MASS	None	10 min@1100 W	[129]
	NaYF ₄ :Yb ³⁺ , R ³⁺ (R ³⁺ : Er, Tm, Yb)	MAH	Solution	5–10 min@60–180 °C	[130]
	KYF ₄ :R ³⁺ (R: Ce, Tb)	MAH	Solution	4 h@200 °C	[131]
	LiYF ₄ :R ³⁺ (R: Eu, Tb, Dy)	MAH	Solution	15 min@140 °C	[44]
Germanate	La ₂ MgGeO ₆ :Mn ⁴⁺	MASS	Carbon	40 min@1000 W	[122]
	Zn ₂ GeO ₄ :Mn ²⁺ , Yb ³⁺	MAH	Solution	Not provided	[132]
Molybdate	CaMoO ₄ :Eu ³⁺ , Dy ³⁺	MASS	Carbon	30 min@560 W	[120]
	CaGd ₂ (MoO ₄) ₄ :Ho ³⁺ , Yb ³⁺	MASG	Solution	30 min@1250 W	[133]
	CaMoO ₄ :Er ³⁺ , Yb ³⁺	MAC	Ethylene Glycol	23 min@1250 W	[134]
	CdMoO ₄ :Eu ³⁺	MAH	Solution	10 min @ 180 °C	[135]
	Cs ₅ Bi(MoO ₄) ₄	MASS		10min@850 W	[136]
	Na ₂ MoO ₄	MASS	Carbon	3 min@840 W (9 cycles)	[121]
	NaY(MoO ₄) ₂ :Yb ³⁺ , Tm ³⁺	MAH	Solution	1 h@180 °C	[137]
	NaCaGd(MoO ₄) ₃ :Ho ³⁺ , Yb ³⁺	MASG	Solution	30 min@1250 W	[138]
	NaCaLa(MoO ₄) ₃	MASG	Solution	30min@1250 W	[139]
	NaCaGd(MoO ₄) ₃ :Er ³⁺ , Yb ³⁺	MASG	Solution	30 min@1250 W	[140]
	NaY(MoO ₄) ₂ :Eu ³⁺	MAH	Solution	10 min@170 °C	[141]
	NaEu(MoO ₄) ₂	MASS	Carbon	3 min@960 W (6 cycles)	[121]
	NaPbLa(MoO ₄) ₃ :Er ³⁺ , Yb ³⁺	MASG	Solution	30 min @ 1250 W	[117]
	PbGd ₂ (MoO ₄) ₄ :Er ³⁺ , Yb ³⁺	MASG	Solution	30 min@1250 W	[142]
	NaLa _{0.95} Eu _{0.05} (MoO ₄) ₂	MASS	Carbon	3 min@960 W (9 cycles)	[121]
	NaLa _{0.95} Pr _{0.025} Dy _{0.025} (MoO ₄) ₂	MASS	Carbon	3 min@960 W (6 cycles)	[121]
	NaSrLa(MoO ₄) ₃ :Er ³⁺ , Yb ³⁺	MASG	Solution	30 min @ 1250 W	[116]
	SrMoO ₄	MAP	Ethylene Glycol	20 min@180 W	[97]
	Nitride	Ca ₂ Si ₅ N ₈ :Eu ²⁺	MASS	None	10 min@100–600 W
Oxide				12 min@1000 W + 10 min@900 W	
	Lu ₂ O ₃ :R ³⁺ (R:Pr, Eu, Tb)	MASS	Carbon		[62]
	Y ₂ O ₃ :Eu ³⁺	US—MA	Water	60 min@150 W	[96]
	SnO ₂ :Eu ³⁺	MAC	Ethylene Glycol	Not Provided	[144]
	TiO ₂ :Sm ³⁺	MAH	Solution	31 min @ 850 W	[145]
	HfO ₂ :Eu ³⁺	MAH	Solution	20 min @ 260 °C	[146]

Table 4. Cont.

Class	Composition	Method	Susceptor	MW Program	Ref.
Oxysulfide	R ₂ O ₂ S:Ti,Mg (R:La, Gd, Y)	MASS	Carbon	10 min@900 W + 15 min@800 W	[76]
	Gd ₂ O ₂ S:Eu ³⁺	MASS	Carbon	2 min@167, 500, and 1000 W 10 min@900 W + 15 min@800 W	[147]
	Gd ₂ O ₂ S:Tb ³⁺	MASS	Carbon	15 min@800 W	[77]
Phosphate	LaPO ₄ :Ce ³⁺ , Tb ³⁺	MAP	Ionic Liquid	10 s@800 W 30 min@80 °C +	[148]
	LaPO ₄ :Eu ³⁺ , Li ⁺	MASG	Gel	30 min@800 °C	[149]
	CePO ₄ :Tb ³⁺	MAP	Solution	4 min@700 W	[150]
	Ca ₃ (PO ₄) ₂ :Gd ³⁺ , Pr ³⁺	MAH	Solution	1 h @ 150 °C (300 W)	[151]
	YPO ₄ :R ³⁺ (R:Eu, Ce, Tb)	MAH	Solution	Not provided	[152]
	Sr ₂ P ₂ O ₇ :Ce ³⁺ , Tb ³⁺	MASS	Solution	15 min @ 1000 °C	[153]
Silicate	Bi ₄ Si ₃ O ₁₂ :Eu ³⁺	MASS	Not provided	15 min @ 750 °C	[154]
	Ca ₂ MgSi ₂ O ₇ :Eu ²⁺	MASS	Carbon	Not provided	[155]
	M ₂ SiO ₄ :Eu ²⁺ (M: Ca, Ba)	MASG	Carbon	10 min + 12 min @ 1000 W	[98]
	Mg ₂ SiO ₄ :R ³⁺ (R: Eu, Tb)	MASG	Solution	12 min @ 800 W 8 min@625 W +	[156]
	Na ₃ YSi ₂ O ₇ :Ce ³⁺	MASS	Carbon	15 min@375 W	[106]
	Sr ₂ SiO ₄ :Tb ³⁺	MASG	Solution	14 min @ 800 W 12 min@1000 W + 10	[157]
	Sr ₂ MgSi ₂ O ₇ :Eu ²⁺ , Dy ³⁺	MASS	Carbon	min@900 W	[57]
	Sr ₃ SiO ₅ :Eu ²⁺	MASS	Carbon	20 min @3200 W	[158]
Sulfide	ZnS:Mn	MAH	Solution	10 min@800 W	[159]
	Zn _{1-x} Cd _x S	MAC	Solution	Not provided	[160]
	ZnS:Pb ²⁺	MASS	None	4 h@ 700 °C 12 min@1000 W + 10	[84]
	SrS:Eu ²⁺ , R ³⁺ (R: Ce, Sm, Er, Dy)	MASS	Carbon	min@900 W	[73]
Titanate	CaTiO ₃ :Eu ³⁺	MASS	Carbon	10–60 min@528–800 W	[161]
	Lu ₂ Ti ₂ O ₇ :Yb ³⁺ , Er ³⁺	MAH	Solution	1 h@200 °C (100 W)	[162]
	BaTiO ₃	MASS	None	1–5 min@2000 W	[163]
Tungstate	CaGd ₂ (WO ₄) ₄ :Er ³⁺ , Yb ³⁺	MASG	Solution	30 min @ 1250 W 20 min@600 °C + 10	[164]
	Li ₃ BaSr(La _{1-x} Eu _x) ₃ (WO ₄) ₈ :Eu ³⁺	MASS	Not provided	min@900 °C	[165]
	Li ₂ Gd ₄ (WO ₄) ₇	MASS	Carbon	10 min@800 °C	[166]
	NaGd(WO ₄) ₂ :Yb ³⁺ , Tm ³⁺ , Ho ³⁺	MAH	Solution	2 h @ 200 °C	[167]
	ZnWO ₄ :Dy ³⁺	MAH	Solution	1 h@140 °C	[100]
	Na ₂ WO ₄	MASS	Carbon	2 min@1080 W (3 cycles)	[121]
	NaEu(WO ₄) ₂	MASS	Carbon	3 min@1080 W (6 cycles)	[121]
	NaLa _{0.95} Eu _{0.05} (WO ₄) ₂	MASS	Carbon	3 min@1080 W (9 cycles)	[121]
	NaLa _{0.95} Pr _{0.025} Dy _{0.025} (WO ₄) ₂	MASS	Carbon	3 min@1080 W (9 cycles)	[121]
	NaY(WO ₄) ₂ : Ho ³⁺ , Yb ³⁺	MAH	Solution	2 h @ 180 °C	[168]
	NaLaMgWO ₆ :Dy ³⁺ , Tm ³⁺	MASS	None	10 min @ 900 °C	[169]
	Vanadate	CeVO ₄	MAH	solution	10 min@600 W
EuVO ₄		MAH	Solution	15 min@150 °C	[99]
Y _{0.5} Eu _{0.5} VO ₄		MAH	Solution	15 min@150 °C	[99]
YP _x V _{1-x} O ₄ :R ³⁺ (R: Eu, Dy, Sm)		MAH	Solution	30 min @ 150 °C	[171]
YVO ₄ :R ³⁺ (R: Eu, Dy, Sm)		MAH	Solution	20 min @ 270 W	[172]

4. Conclusions and Outlook

In conclusion, the future of luminescent materials obtained by microwave methodologies is bright yet challenging. The microwave-assisted synthesis methodologies are incredibly versatile and are proved to reduce energy consumption and considerably lower the synthesis time of inorganic luminescent materials. Not only is this method more cost-

effective than its conventional heating counterpart, but it also has an additional effect on the synthesis, leading to increased reaction rates due to the enhanced ionic diffusion promoted by the electromagnetic field of the MW. Dielectric heating is also an excellent strategy to obtain homogeneous luminescent materials, ideal for illumination purposes, once the intense local heat generation leads to a slight temperature gradient. The rapid heating and quenching of microwave-assisted heating can induce formation of structural defects that allow for high storage capacity of the persistent phosphors.

The shortcoming of the microwave methodologies is sometimes the lack of quantitative comparison between the microwave-obtained materials and other methods, making the advantages challenging to interpret. Furthermore, the methodologies, especially the solid-state one, need to be consolidated in the luminescence community as a recognized method for obtaining phosphor materials. If the radiation hazards are appropriately controlled, novel customized setups can be designed for other methods, such as reflux synthesis using organic solvents and ionic liquids, that might allow obtaining nanoparticles with controlled morphology, such as NaYF₄. The design of a dedicated microwave irradiation reactor cavity could also open new avenues for luminescent materials, leading to possibilities of temperature and atmosphere control.

The combination of these properties makes microwave-assisted synthesis an excellent method to obtain luminescent materials that benefit from high crystallinity, well-dispersed dopants, and homogeneous structures. These properties translate to high quantum yields, high brightness, and large storage capacity. Moreover, the microwave-assisted methods allow for fast screening for compositional tailoring, a widespread strategy when tuning the color emission of crystal field-sensitive ions, such as Eu²⁺ and Ce³⁺.

Author Contributions: Conceptualization, J.M.d.C., C.C.S.P., M.S.d.N.S., M.C.F.C.F., H.F.d.B.; writing—original draft preparation, J.M.d.C.; writing—review and editing, J.M.d.C., C.C.S.P., M.S.d.N.S., M.C.F.C.F., H.F.d.B.; supervision, J.M.d.C.; project administration, J.M.d.C. All authors have read and agreed to the published version of the manuscript.

Funding: Project FAPESP n° 2017/05195-5, and FAPESP n° 2020/10228-2.

Institutional Review Board Statement: Non-applicable.

Informed Consent Statement: Non-applicable.

Conflicts of Interest: The authors declare no conflict of interest.

References

1. Blasse, G.; Grabmaier, B.C. *Luminescent Materials*; Springer: Berlin/Heidelberg, Germany, 1994; ISBN 978-3-540-58019-5.
2. Ronda, C. Rare-Earth Phosphors: Fundamentals and Applications. In *Reference Module in Materials Science and Materials Engineering*; Elsevier: Amsterdam, The Netherlands, 2017; ISBN 9780128035818.
3. Malta, O.L.; Brito, H.F.; Menezes, J.F.S.; e Silva, F.R.G.; Alves, S.; Farias, F.S.; de Andrade, A.V.M. Spectroscopic properties of a new light-converting device Eu(thenoyltrifluoroacetate)₃ 2(dibenzyl sulfoxide). A theoretical analysis based on structural data obtained from a sparkle model. *J. Lumin.* **1997**, *75*, 255–268. [[CrossRef](#)]
4. Rodrigues, L.C.V. Preparação e desenvolvimento do mecanismo da luminescência persistente de materiais dopados com íons terras raras. Doctoral Thesis, University of São Paulo, Butanta, Brazil, 2012.
5. Verstraete, R.; Sijbom, H.F.; Joos, J.J.; Korthout, K.; Poelman, D.; Detavernier, C.; Smet, P.F. Red Mn⁴⁺-Doped Fluoride Phosphors: Why Purity Matters. *ACS Appl. Mater. Interfaces* **2018**, *10*, 18845–18856. [[CrossRef](#)]
6. da Silva, M.N.; Carvalho, J.M.; de Abreu Fantini, M.C.; Chiavacci, L.A.; Bourgaux, C. Nanosized ZnGa₂O₄:Cr₃₊ Spinel as Highly Luminescent Materials for Bioimaging. *ACS Appl. Nano Mater.* **2019**, *2*, 6918–6927. [[CrossRef](#)]
7. Ming, H.; Liu, S.; Liu, L.; Peng, J.; Fu, J.; Du, F.; Ye, X. Highly Regular, Uniform K₃ScF₆:Mn⁴⁺ Phosphors: Facile Synthesis, Microstructures, Photoluminescence Properties, and Application in Light-Emitting Diode Devices. *ACS Appl. Mater. Interfaces* **2018**, *10*, 19783–19795. [[CrossRef](#)] [[PubMed](#)]
8. Aarts, L.; van der Ende, B.; Reid, M.F.; Meijerink, A. Downconversion for solar cells in YF₃:Pr³⁺, Yb³⁺. *Spectrosc. Lett.* **2010**, *43*, 373–381. [[CrossRef](#)]
9. Terra, I.A.A.; Borrero-González, L.J.; Carvalho, J.M.; Terrile, M.C.; Felinto, M.C.F.C.; Brito, H.F.; Nunes, L.A.O. Spectroscopic properties and quantum cutting in Tb³⁺–Yb³⁺ co-doped ZrO₂ nanocrystals. *J. Appl. Phys.* **2013**, *113*, 073105. [[CrossRef](#)]
10. Vergeer, P.; Vlugt, T.J.H.; Kox, M.H.F.; Den Hertog, M.I.; Van Der Herden, J.P.J.M.; Meijerink, A. Quantum cutting by cooperative energy transfer in Yb_xY_{1-x}PO₄:Tb³⁺. *Phys. Rev. B Condens. Matter Mater. Phys.* **2005**, *71*, 1–11. [[CrossRef](#)]

11. Yi, G.S.; Chow, G.M. Synthesis of hexagonal-phase NaYF₄:Yb,Er and NaYF₄:Yb,Tm nanocrystals with efficient up-conversion fluorescence. *Adv. Funct. Mater.* **2006**, *16*, 2324–2329. [[CrossRef](#)]
12. Hu, L.; Wang, P.; Zhao, M.; Liu, L.; Zhou, L.; Li, B.; Albaqami, F.H.; El-Toni, A.M.; Li, X.; Xie, Y.; et al. Near-infrared rechargeable “optical battery” implant for irradiation-free photodynamic therapy. *Biomaterials* **2018**, *163*, 154–162. [[CrossRef](#)] [[PubMed](#)]
13. Auzel, F. Upconversion and Anti-Stokes Processes with f and d Ions in Solids. *Chem. Rev.* **2004**, *104*, 139–173. [[CrossRef](#)]
14. Smits, K.; Grigorjeva, L.; Millers, D.; Sarakovskis, A.; Opalinska, A.; Fidelus, J.D.; Lojkowski, W. Europium doped zirconia luminescence. *Opt. Mater.* **2010**, *32*, 827–831. [[CrossRef](#)]
15. Yang, C.; Fu, L.M.; Wang, Y.; Zhang, J.P.; Wong, W.T.; Ai, X.C.; Qiao, Y.F.; Zou, B.S.; Gui, L.L. A highly luminescent europium complex showing visible-light-sensitized red emission: Direct observation of the singlet pathway. *Angew. Chemie. Int. Ed.* **2004**, *43*, 5010–5013. [[CrossRef](#)]
16. Pappalardo, R. Calculated quantum yields for photon-cascade emission (PCE) for Pr³⁺ and Tm³⁺ in fluoride hosts. *J. Lumin.* **1976**, *14*, 159–193. [[CrossRef](#)]
17. Piper, W.W.; DeLuca, J.A.; Ham, F.S. Cascade fluorescent decay in Pr³⁺-doped fluorides: Achievement of a quantum yield greater than unity for emission of visible light. *J. Lumin.* **1974**, *8*, 344–348. [[CrossRef](#)]
18. Wegh, R.T.; Donker, H.; Oskam, K.D.; Meijerink, A. Visible quantum cutting in LiGdF₄:Eu³⁺ through downconversion. *Science* **1999**, *283*, 663–666. [[CrossRef](#)] [[PubMed](#)]
19. Wegh, R.; Donker, H.; van Loef, E.V.; Oskam, K.; Meijerink, A. Quantum cutting through downconversion in rare-earth compounds. *J. Lumin.* **2000**, *87–89*, 1017–1019. [[CrossRef](#)]
20. Sommerdijk, J.L.; Bril, A.; de Jager, A.W. Two photon luminescence with ultraviolet excitation of trivalent praseodymium. *J. Lumin.* **1974**, *8*, 341–343. [[CrossRef](#)]
21. Bloembergen, N. Solid State Infrared Quantum Counters. *Phys. Rev. Lett.* **1959**, *2*, 84–85. [[CrossRef](#)]
22. Zhou, B.; Shi, B.; Jin, D.; Liu, X. Controlling upconversion nanocrystals for emerging applications. *Nat. Nanotechnol.* **2015**, *10*, 924–936. [[CrossRef](#)]
23. Xu, J.; Tanabe, S. Persistent luminescence instead of phosphorescence: History, mechanism, and perspective. *J. Lumin.* **2019**, *205*, 581–620. [[CrossRef](#)]
24. Van den Eeckhout, K.; Poelman, D.; Smet, P.F. Persistent luminescence in Non-Eu²⁺-Doped compounds: A review. *Materials* **2013**, *6*, 2789. [[CrossRef](#)]
25. Van den Eeckhout, K.; Smet, P.F.; Poelman, D. Persistent Luminescence in Eu²⁺-Doped Compounds: A Review. *Materials* **2010**, *3*, 2536–2566. [[CrossRef](#)]
26. Im, W.B.; George, N.; Kurzman, J.; Brinkley, S.; Mikhailovsky, A.; Hu, J.; Chmelka, B.F.; Denbaars, S.P.; Seshadri, R. Efficient and color-tunable oxyfluoride solid solution phosphors for solid-state white lighting. *Adv. Mater.* **2011**, *23*, 2300–2305. [[CrossRef](#)] [[PubMed](#)]
27. Im, W.B.; Brinkley, S.; Hu, J.; Mikhailovsky, A.; Denbaars, S.P.; Seshadri, R. Sr_{2.975-x}Ba_xCe_{0.025}AlO₄F: A highly efficient green-emitting oxyfluoride phosphor for solid state white lighting. *Chem. Mater.* **2010**, *22*, 2842–2849. [[CrossRef](#)]
28. Dorenbos, P. Ce³⁺ 5d-centroid shift and vacuum referred 4f-electron binding energies of all lanthanide impurities in 150 different compounds. *J. Lumin.* **2013**, *135*, 93–104. [[CrossRef](#)]
29. Yamaga, M.; Yosida, T.; Hara, S.; Kodama, N.; Henderson, B. Optical and electron spin resonance spectroscopy of Ti³⁺ and Ti⁴⁺ in Al₂O₃. *J. Appl. Phys.* **1994**, *75*, 1111–1117. [[CrossRef](#)]
30. Katumo, N.; Gao, G.; Laufer, F.; Richards, B.S.; Howard, I.A. Smartphone-Based Luminescent Thermometry via Temperature-Sensitive Delayed Fluorescence from Gd₂O₃:Eu³⁺. *Adv. Opt. Mater.* **2020**, *8*, 2000507. [[CrossRef](#)]
31. Chen, R.; Pagonis, V. Modelling thermal activation characteristics of the sensitization of thermoluminescence in quartz. *J. Phys. D Appl. Phys.* **2004**, *37*, 159–164. [[CrossRef](#)]
32. Chen, R.; Pagonis, V.; Lawless, J.L. Evaluated thermoluminescence trapping parameters—What do they really mean? *Radiat. Meas.* **2016**, *91*, 21–27. [[CrossRef](#)]
33. Kristianpoller, N.; Chen, R.; Israeli, M. Dose dependence of thermoluminescence peaks. *J. Phys. D Appl. Phys.* **1974**, *7*, 1063–1072. [[CrossRef](#)]
34. Dong, H.; Du, S.R.; Zheng, X.Y.; Lyu, G.M.; Sun, L.D.; Li, L.D.; Zhang, P.Z.; Zhang, C.; Yan, C.H. Lanthanide Nanoparticles: From Design toward Bioimaging and Therapy. *Chem. Rev.* **2015**, *115*, 10725–10815. [[CrossRef](#)]
35. Wang, Q.; Liu, N.; Hou, Z.; Shi, J.; Su, X.; Sun, X. Radioiodinated Persistent Luminescence Nanoplatform for Radiation-Induced Photodynamic Therapy and Radiotherapy. *Adv. Healthc. Mater.* **2021**, *10*, 2000802. [[CrossRef](#)] [[PubMed](#)]
36. Liu, J.; Lécuyer, T.; Seguin, J.; Mignet, N.; Scherman, D.; Viana, B.; Richard, C. Imaging and therapeutic applications of persistent luminescence nanomaterials. *Adv. Drug Deliv. Rev.* **2019**, *138*, 193–210. [[CrossRef](#)]
37. Tan, H.; Wang, T.; Shao, Y.; Yu, C.; Hu, L. Crucial Breakthrough of Functional Persistent Luminescence Materials for Biomedical and Information Technological Applications. *Front. Chem.* **2019**, *7*. [[CrossRef](#)]
38. Smet, P.F.; Parmentier, A.B.; Poelman, D. Selecting Conversion Phosphors for White Light-Emitting Diodes. *J. Electrochem. Soc.* **2011**, *158*, R37. [[CrossRef](#)]
39. Schubert, E.F.; Kim, J.K. Solid-state light sources getting smart. *Science* **2005**, *308*, 1274–1278. [[CrossRef](#)]
40. George, N.C.; Denault, K.A.; Seshadri, R. Phosphors for Solid-State White Lighting. *Annu. Rev. Mater. Res.* **2013**, *43*, 481–501. [[CrossRef](#)]

41. Sijbom, H.F.; Verstraete, R.; Joos, J.J.; Poelman, D.; Smet, P.F. $\text{K}_2\text{SiF}_6:\text{Mn}^{4+}$ as a red phosphor for displays and warm-white LEDs: A review of properties and perspectives. *Opt. Mater. Express* **2017**, *7*, 3332. [[CrossRef](#)]
42. Saradhi, M.P.; Boudin, S.; Varadaraju, U.V.; Raveau, B. A new $\text{BaB}_2\text{Si}_2\text{O}_8:\text{Eu}^{2+}/\text{Eu}^{3+}, \text{Tb}^{3+}$ phosphor Synthesis and photoluminescence properties. *J. Solid State Chem.* **2010**, *183*, 2496–2500. [[CrossRef](#)]
43. Hizal Abacı, Ö.C.; Mete, E.; Esenturk, O.; Yılmaz, A. Tunable optical properties and DFT calculations of RE^{3+} codoped LaBO_3 phosphors. *Opt. Mater.* **2019**, *98*, 3–12. [[CrossRef](#)]
44. Andrade, A.B.; Bispo, G.F.C.; Macedo, Z.S.; Valerio, M.E.G. Synthesis and characterization of luminescent Ln^{3+} (Ln = Eu, Tb and Dy)-doped LiYF_4 microcrystals produced by a facile microwave-assisted hydrothermal method. *J. Lumin.* **2020**, *219*, 116843. [[CrossRef](#)]
45. Chuang, Y.J.; Zhen, Z.; Zhang, F.; Liu, F.; Mishra, J.P.; Tang, W.; Chen, H.; Huang, X.; Wang, L.; Chen, X.; et al. Photostimulable near-infrared persistent luminescent nanoprobe for ultrasensitive and longitudinal deep-tissue bio-imaging. *Theranostics* **2014**, *4*, 1112–1122. [[CrossRef](#)]
46. Kang, F.; Sun, G.; Boutinaud, P.; Wu, H.; Ma, F.-X.; Lu, J.; Gan, J.; Bian, H.; Gao, F.; Xiao, S. Recent advances and prospects of persistent luminescent materials as inner secondary self-luminous light source for photocatalytic applications. *Chem. Eng. J.* **2021**, *403*, 126099. [[CrossRef](#)]
47. Van der Heggen, D.; Joos, J.J.; Rodríguez Burbano, D.C.; Capobianco, J.A.; Smet, P.F. Counting the photons: Determining the absolute storage capacity of persistent phosphors. *Materials* **2017**, *10*, 867. [[CrossRef](#)]
48. Van der Heggen, D.; Joos, J.; Smet, P.F. On the importance of evaluating the intensity dependency of the quantum efficiency: Impact on LEDs and persistent phosphors. *ACS Photonics* **2018**, *5*, 4529–4537. [[CrossRef](#)]
49. Bhattacharya, M.; Basak, T. A review on the susceptor assisted microwave processing of materials. *Energy* **2016**, *97*, 306–338. [[CrossRef](#)]
50. Schiffmann, R.F. Microwave and dielectric drying. In *Handbook of Industrial Drying*, 4th ed.; CRC Press: Boca Raton, FL, USA, 2014; ISBN 9781466596665.
51. Kitchen, H.J.; Vallance, S.R.; Kennedy, J.L.; Tapia-Ruiz, N.; Carassiti, L.; Harrison, A.; Whittaker, A.G.; Drysdale, T.D.; Kingman, S.W.; Gregory, D.H. Modern microwave methods in solid-state inorganic materials chemistry: From fundamentals to manufacturing. *Chem. Rev.* **2014**, *114*, 1170–1206. [[CrossRef](#)]
52. Rao, K.J.; Ramesh, P.D. Use of microwaves for the synthesis and processing of materials. *Bull. Mater. Sci.* **1995**, *18*, 447–465. [[CrossRef](#)]
53. Naitoh, K.; Takizawa, T.; Matsuse, T. Controlled microwave irradiation for the synthesis of $\text{YBa}_2\text{Cu}_3\text{O}_{7-x}$ superconductors. *Jpn. J. Appl. Phys. Part 2 Lett.* **1999**. [[CrossRef](#)]
54. Vaidhyanathan, B.; Balaji, K.; Rao, K.J. Microwave-Assisted Solid-State Synthesis of Oxide Ion Conducting Stabilized Bismuth Vanadate Phases. *Chem. Mater.* **1998**, *10*, 3400–3404. [[CrossRef](#)]
55. Rao, K.J.; Vaidhyanathan, B.; Ganguli, M.; Ramakrishnan, P.A. Synthesis of inorganic solids using microwaves. *Chem. Mater.* **1999**, *11*, 882–895. [[CrossRef](#)]
56. Kappe, C.O. Controlled microwave heating in modern organic synthesis. *Angew. Chem. Int. Ed.* **2004**, *43*, 6250–6284. [[CrossRef](#)]
57. Miranda De Carvalho, J.; Van Der Heggen, D.; Martin, L.I.D.J.; Smet, P.F. Microwave-assisted synthesis followed by a reduction step: Making persistent phosphors with a large storage capacity. *Dalt. Trans.* **2020**, *49*, 4518–4527. [[CrossRef](#)] [[PubMed](#)]
58. Sebastian, M.T. *Dielectric Materials for Wireless Communication*; Elsevier: Amsterdam, The Netherlands, 2008; ISBN 9780080453309.
59. Hotta, M.; Hayashi, M.; Lanagan, M.T.; Agrawal, D.K.; Nagata, K. Complex permittivity of graphite, carbon black and coal powders in the ranges of X-band frequencies (8.2 to 12.4 GHz) and between 1 and 10 GHz. *ISIJ Int.* **2011**, *51*, 1766–1772. [[CrossRef](#)]
60. Balanis, C.A. (Ed.) *Advanced Engineering Electromagnetics*; Wiley: Hoboken, NJ, USA, 2007; ISBN 9780470589489.
61. Gupta, M.; Eugene, W.W.L. *Microwaves and Metals*; John Wiley & Sons: Hoboken, NJ, USA, 2011; ISBN 9780470822722.
62. Pedroso, C.C.S.; Carvalho, J.M.; Rodrigues, L.C.V.; Hölsä, J.; Brito, H.F. Rapid and Energy-Saving Microwave-Assisted Solid-State Synthesis of Pr^{3+} -, Eu^{3+} -, or Tb^{3+} -Doped Lu_2O_3 Persistent Luminescence Materials. *ACS Appl. Mater. Interfaces* **2016**, *8*, 19593–19604. [[CrossRef](#)] [[PubMed](#)]
63. Levin, E.E.; Grebenkemper, J.H.; Pollock, T.M.; Seshadri, R. Protocols for High Temperature Assisted-Microwave Preparation of Inorganic Compounds. *Chem. Mater.* **2019**, *31*, 7151–7159. [[CrossRef](#)]
64. Bykov, Y.V.; Rybakov, K.I.; Semenov, V.E. High-temperature microwave processing of materials. *J. Phys. D Appl. Phys.* **2001**, *34*, R55–R75. [[CrossRef](#)]
65. Ku, H.S.; Ball, J.A.R.; Siores, E. Review-Microwave Processing of Materials: Part III. *HKIE Trans.* **2001**, *8*, 44–50. [[CrossRef](#)]
66. Metaxas, A.C.; Meredith, R.J. Dielectric loss. In *Industrial Microwave Heating*; IET: Six Hills Way, UK, 2011; pp. 5–25.
67. Li, W.; Adlung, M.; Zhang, Q.; Wickleder, C.; Schmedt auf der Günne, J. A Guide to Brighter Phosphors-Linking Luminescence Properties to Doping Homogeneity Probed by NMR. *ChemPhysChem* **2019**, *20*, 3245–3250. [[CrossRef](#)]
68. Clark, D.E.; Folz, D.C.; West, J.K. Processing materials with microwave energy. *Mater. Sci. Eng. A* **2000**, *287*, 153–158. [[CrossRef](#)]
69. Gusarov, V.V. Fast Solid-Phase Chemical Reactions. *Russ. J. Gen. Chem.* **1997**, *67*, 1846–1851.
70. Whittaker, A.G. Diffusion in microwave-heated ceramics. *Chem. Mater.* **2005**, *17*, 3426–3432. [[CrossRef](#)]
71. Vanetsev, A.S.; Baranchikov, A.E.; Tret'yakov, Y.D. Kinetics of microwave-enhanced solid-phase reaction of NiFe_2O_4 formation. *Russ. J. Inorg. Chem.* **2008**, *53*, 495–498. [[CrossRef](#)]

72. Rödiger, K.; Dreyer, K.; Gerdes, T.; Willert-Porada, M. Microwave sintering of hardmetals. *Int. J. Refract. Met. Hard Mater.* **1998**, *16*, 409–416. [[CrossRef](#)]
73. dos Santos, D.O.A.; Giordano, L.; Barbará, M.A.S.G.; Portes, M.C.; Pedroso, C.C.S.; Teixeira, V.C.; Lastusaari, M.; Rodrigues, L.C.V. Abnormal co-doping effect on the red persistent luminescence SrS:Eu²⁺,RE³⁺ materials. *Dalt. Trans.* **2020**, *49*, 16386–16393. [[CrossRef](#)]
74. Chandramouli, V.; Anthonysamy, S.; Vasudeva Rao, P.R.; Divakar, R.; Sundararaman, D. Microwave synthesis of solid solutions of urania and thoria-A comparative study. *J. Nucl. Mater.* **1998**, *254*, 55–64. [[CrossRef](#)]
75. Ganesh, I.; Johnson, R.; Rao, G.V.N.; Mahajan, Y.R.; Madavendra, S.S.; Reddy, B.M. Microwave-assisted combustion synthesis of nanocrystalline MgAl₂O₄ spinel powder. *Ceram. Int.* **2005**, *31*, 67–74. [[CrossRef](#)]
76. Miranda De Carvalho, J.; Pedroso, C.C.S.; Machado, I.P.; Hölsä, J.; Rodrigues, L.C.V.; Gluchowski, P.; Lastusaari, M.; Brito, H.F. Persistent luminescence warm-light LEDs based on Ti-doped RE₂O₂S materials prepared by rapid and energy-saving microwave-assisted synthesis. *J. Mater. Chem. C* **2018**, *6*, 8897–8905. [[CrossRef](#)]
77. Machado, I.P.; Teixeira, V.C.; Pedroso, C.C.S.; Brito, H.F.; Rodrigues, L.C.V. X-ray scintillator Gd₂O₂S:Tb³⁺ materials obtained by a rapid and cost-effective microwave-assisted solid-state synthesis. *J. Alloys Compd.* **2019**, *777*, 638–645. [[CrossRef](#)]
78. Lee, W.C.; Liu, K.S.; Lin, I.N. Microwave sintering Pb(Zr_{0.52}Ti_{0.48})O₃ piezoelectric ceramics. *Ferroelectrics* **2001**, *262*, 293–298. [[CrossRef](#)]
79. Walkiewicz, J.W.; Kazonich, G.; McGill, S.L. Microwave Heating Characteristics of Selected Minerals and Compounds. *Miner. Metall. Process.* **1988**, *5*, 39–42. [[CrossRef](#)]
80. Song, B.; Zhao, B.; Fan, L.; Fan, B.; Wang, H.; Guo, X.; Zhang, R. Investigation on heating behavior during the preparation of SiC crystals by microwave sintering. *Int. J. Appl. Ceram. Technol.* **2017**, *14*, 880–888. [[CrossRef](#)]
81. Rao, K.V.; Smakula, A. Dielectric properties of cobalt oxide, nickel oxide, and their mixed crystals. *J. Appl. Phys.* **1965**, *36*, 2031–2038. [[CrossRef](#)]
82. Kulesza, D.; Bolek, P.; Bos, A.J.J.; Zych, E. Lu₂O₃-based storage phosphors. An (in)harmonious family. *Coord. Chem. Rev.* **2016**, *325*, 29–40. [[CrossRef](#)]
83. Birkel, A.; Denault, K.A.; George, N.C.; Doll, C.E.; Héry, B.; Mikhailovsky, A.A.; Birkel, C.S.; Hong, B.C.; Seshadri, R. Rapid microwave preparation of highly efficient Ce³⁺-substituted garnet phosphors for solid state white lighting. *Chem. Mater.* **2012**, *24*, 1198–1204. [[CrossRef](#)]
84. Game, D.N.; Palan, C.B.; Ingale, N.B.; Omanwar, S.K. Microwave-assisted synthesis and photoluminescence properties of ZnS:Pb²⁺ nanophosphor for solid-state lighting. *Bull. Mater. Sci.* **2017**, *40*, 1441–1445. [[CrossRef](#)]
85. Kamińska, I.; Jankowski, D.; Sikora, B.; Kowalik, P.; Minikayev, R.; Wojciechowski, T.; Chojnacki, M.; Sobczak, K.; Rybusiński, J.; Szczytko, J.; et al. Structural, optical and magnetic properties of Y_{3–0.02–x}Er_{0.02}Yb_xAl₅O₁₂ (0 <x <0.20) nanocrystals: Effect of Yb content. *Nanotechnology* **2020**, *31*, 225711. [[CrossRef](#)]
86. Liu, G.; Li, J.; Chen, K. Combustion synthesis of refractory and hard materials: A review. *Int. J. Refract. Met. Hard Mater.* **2013**, *39*, 90–102. [[CrossRef](#)]
87. Patil, K.C.; Aruna, S.T.; Ekambaram, S. Combustion synthesis. *Curr. Opin. Solid State Mater. Sci.* **1997**. [[CrossRef](#)]
88. Patil, K.C.; Aruna, S.T.; Mimani, T. Combustion synthesis: An update. *Curr. Opin. Solid State Mater. Sci.* **2002**, *6*, 507–512. [[CrossRef](#)]
89. Varma, A.; Rogachev, A.S.; Mukasyan, A.S.; Hwang, S. Combustion Synthesis of Advanced Materials: Principles and Applications. In *Advances in Chemical Engineering*; Academic Press: Cambridge, MA, USA, 1998; pp. 79–226.
90. Kamińska, I.; Fronc, K.; Sikora, B.; Mouawad, M.; Siemiarczuk, A.; Szewczyk, M.; Sobczak, K.; Wojciechowski, T.; Zaleszczyk, W.; Minikayev, R.; et al. Upconverting/magnetic: Gd₂O₃:(Er³⁺,Yb³⁺,Zn²⁺) nanoparticles for biological applications: Effect of Zn²⁺ doping. *RSC Adv.* **2015**, *5*, 78361–78373. [[CrossRef](#)]
91. Meng, L.-Y.; Wang, B.; Ma, M.-G.; Lin, K.-L. The progress of microwave-assisted hydrothermal method in the synthesis of functional nanomaterials. *Mater. Today Chem.* **2016**, *1–2*, 63–83. [[CrossRef](#)]
92. Johnson, G.M.; Mead, P.J.; Weller, M.T. Synthesis of a range of anion-containing gallium and germanium sodalites. *Microporous Mesoporous Mater.* **2000**, *38*, 445–460. [[CrossRef](#)]
93. Misch, L.M.; Birkel, A.; Figg, C.A.; Fors, B.P.; Hawker, C.J.; Stucky, G.D.; Seshadri, R. Rapid microwave-assisted sol-gel preparation of Pd-substituted LnFeO₃ (Ln = Y, La): Phase formation and catalytic activity. *Dalt. Trans.* **2014**, *43*, 2079–2087. [[CrossRef](#)]
94. Teston, E.; Richard, S.; Maldiney, T.; Lièvre, N.; Wang, G.Y.; Motte, L.; Richard, C.; Lalatonne, Y. Non-aqueous sol-gel synthesis of ultra small persistent luminescence nanoparticles for near-infrared in vivo imaging. *Chem. A Eur. J.* **2015**, *21*, 7350–7354. [[CrossRef](#)]
95. Bilecka, I.; Djerdj, I.; Niederberger, M. One-minute synthesis of crystalline binary and ternary metal oxide nanoparticles. *Chem. Commun.* **2008**, *7*, 886–888. [[CrossRef](#)] [[PubMed](#)]
96. Bi, J.; Sun, L.; Wei, Q.; Zhang, K.; Zhu, L.; Wei, S.; Liao, D.; Sun, J. Rapid ultrasonic-microwave assisted synthesis of Eu³⁺ doped Y₂O₃ nanophosphors with enhanced luminescence properties. *J. Mater. Res. Technol.* **2020**, *9*, 9523–9530. [[CrossRef](#)]
97. Thongtem, T.; Phuruangrat, A.; Thongtem, S. Microwave-assisted synthesis and characterization of SrMoO₄ and SrWO₄ nanocrystals. *J. Nanoparticle. Res.* **2010**, *12*, 2287–2294. [[CrossRef](#)]

98. Birkel, A.; DeCino, N.A.; George, N.C.; Hazelton, K.A.; Hong, B.-C.; Seshadri, R. Eu²⁺-doped M₂SiO₄ (M = Ca, Ba) phosphors prepared by a rapid microwave-assisted sol-gel method: Phase formation and optical properties. *Solid State Sci.* **2013**, *19*, 51–57. [[CrossRef](#)]
99. Duée, N.; Ambard, C.; Pereira, F.; Portehault, D.; Viana, B.; Vallé, K.; Autissier, D.; Sanchez, C. New Synthesis Strategies for Luminescent YVO₄:Eu and EuVO₄ Nanoparticles with H₂O₂ Selective Sensing Properties. *Chem. Mater.* **2015**, *27*, 5198–5205. [[CrossRef](#)]
100. Gonçalves, R.F.; Longo, E.; Marques, A.P.A.; Silva, M.D.P.; Cavalcante, L.S.; Nogueira, I.C.; Pinatti, I.M.; Pereira, P.F.S.; Godinho, M.J. Structural investigation and photoluminescent properties of ZnWO₄:Dy₃₊ nanocrystals. *J. Mater. Sci. Mater. Electron.* **2017**, *28*, 15466–15479. [[CrossRef](#)]
101. Zych, E.; Dereń, P.; Stręk, W.; Meijerink, A.; Mielcarek, W.; Domagala, K. Preparation, X-ray analysis and spectroscopic investigation of nanostructured Lu₂O₃:Tb. *J. Alloys Compd.* **2001**, *323–324*, 8–12. [[CrossRef](#)]
102. Trojan-Piegza, J.; Zych, E. Afterglow Luminescence of Lu₂O₃:Eu Ceramics Synthesized at Different Atmospheres. *J. Phys. Chem. C* **2010**, *114*, 4215–4220. [[CrossRef](#)]
103. Matsuzawa, T.; Aoki, Y.; Takeuchi, N.; Murayama, Y. A New Long Phosphorescent Phosphor with High Brightness, SrAl₂O₄:Eu²⁺, Dy³⁺. *J. Electrochem. Soc.* **1996**, *143*, 2670–2673. [[CrossRef](#)]
104. Blasse, G.; Bril, A. A new phosphor for flying-spot cathode-ray tubes for color television: Yellow-emitting Y₃Al₅O₁₂-Ce³⁺. *Appl. Phys. Lett.* **1967**, *11*, 53–55. [[CrossRef](#)]
105. Finley, E.; Paterson, A.S.; Cobb, A.; Willson, R.C.; Brgoch, J. Reducing particle size of persistent luminescent SrAl₂O₄:Eu²⁺, Dy³⁺ via microwave-assisted, reverse micelle synthesis. *Opt. Mater. Express* **2017**, *7*, 2597. [[CrossRef](#)]
106. Brgoch, J.; Borg, C.K.H.; Denault, K.A.; Douglas, J.R.; Amanda Strom, T.; DenBaars, S.P.; Seshadri, R. Rapid microwave preparation of cerium-substituted sodium yttrium silicate phosphors for solid state white lighting. *Solid State Sci.* **2013**, *26*, 115–120. [[CrossRef](#)]
107. Norrbo, I.; Gluchowski, P.; Hyppänen, I.; Laihininen, T.; Laukkanen, P.; Mäkelä, J.; Mamedov, F.; Santos, H.S.; Sinkkonen, J.; Tuomisto, M.; et al. Mechanisms of Tenebrescence and Persistent Luminescence in Synthetic Hackmanite Na₈Al₆Si₆O₂₄(Cl,S)₂. *ACS Appl. Mater. Interfaces* **2016**, *8*, 11592–11602. [[CrossRef](#)] [[PubMed](#)]
108. Agamah, C.; Vuori, S.; Colinet, P.; Norrbo, I.; Miranda de Carvalho, J.; Okada Nakamura, L.; Lindblom, J.; van Goethem, L.; Emmermann, A.; Saarinen, T.; et al. Hackmanite—The Natural Glow-in-the-Dark Material. *Chem. Mater.* **2020**, *32*, 8895–8905. [[CrossRef](#)]
109. Norrbo, I.; Carvalho, J.M.; Laukkanen, P.; Mäkelä, J.; Mamedov, F.; Peurla, M.; Helminen, H.; Pihlasalo, S.; Härmä, H.; Sinkkonen, J. Lanthanide and Heavy Metal Free Long White Persistent Luminescence from Ti Doped Li-Hackmanite: A Versatile, Low-Cost Material. *Adv. Funct. Mater.* **2017**, *27*, 1696547. [[CrossRef](#)]
110. Norrbo, I.; Gluchowski, P.; Paturi, P.; Sinkkonen, J.; Lastusaari, M. Persistent Luminescence of Tenebrescent Na₈Al₆Si₆O₂₄(Cl,S)₂: Multifunctional Optical Markers. *Inorg. Chem.* **2015**, *54*, 7717–7724. [[CrossRef](#)] [[PubMed](#)]
111. Carvalho, J.M.; Norrbo, I.; Ando, R.A.; Brito, H.F.; Fantini, M.C.A.; Lastusaari, M. Fast, low-cost preparation of hackmanite minerals with reversible photochromic behavior using a microwave-assisted structure-conversion method. *Chem. Commun.* **2018**, *54*, 7326–7329. [[CrossRef](#)]
112. Brgoch, J.; Kloß, S.D.; Denault, K.A.; Seshadri, R. Accessing (Ba_{1-x} Sr_x)Al₂Si₂O₈:Eu Phosphors for Solid State White Lighting via Microwave-assisted Preparation: Tuning Emission Color by Coordination Environment. *Zeitschrift für Anorg. Allg. Chem.* **2014**, *640*, 1182–1189. [[CrossRef](#)]
113. Machida, M.; Kawano, T.; Eto, M.; Zhang, D.; Ikeue, K. Ln dependence of the large-capacity oxygen storage/release property of Ln oxysulfate/oxysulfide systems. *Chem. Mater.* **2007**, *19*, 954–960. [[CrossRef](#)]
114. Zhang, J.; Wang, X.; Zhang, X.; Zhao, X.; Liu, X.; Peng, L. Microwave synthesis of NaLa(MoO₄)₂ microcrystals and their near-infrared luminescent properties with lanthanide ion doping (Er³⁺, Nd³⁺, Yb³⁺). *Inorg. Chem. Commun.* **2011**, *14*, 1723–1727. [[CrossRef](#)]
115. Lim, C.S.; Aleksandrovsky, A.S.; Molokeyev, M.S.; Oreshonkov, A.S.; Ikonnikov, D.A.; Atuchin, V.V. Triple molybdate scheelite-type upconversion phosphor NaCaLa(MoO₄)₃:Er³⁺/Yb³⁺: Structural and spectroscopic properties. *Dalt. Trans.* **2016**, *45*, 15541–15551. [[CrossRef](#)]
116. Lim, C.S.; Aleksandrovsky, A.S.; Molokeyev, M.S.; Oreshonkov, A.S.; Atuchin, V.V. Microwave synthesis and spectroscopic properties of ternary scheelite-type molybdate phosphors NaSrLa(MoO₄)₃:Er³⁺, Yb³⁺. *J. Alloys Compd.* **2017**, *713*, 156–163. [[CrossRef](#)]
117. Lim, C.S.; Aleksandrovsky, A.S.; Atuchin, V.V.; Molokeyev, M.S.; Oreshonkov, A.S. Microwave sol-gel synthesis, microstructural and spectroscopic properties of scheelite-type ternary molybdate upconversion phosphor NaPbLa(MoO₄)₃:Er³⁺/Yb³⁺. *J. Alloys Compd.* **2020**, *826*, 152095. [[CrossRef](#)]
118. Klinbumrung, A.; Phuruangrat, A.; Thongtem, T.; Thongtem, S. Synthesis, Characterization and Optical Properties of BaMoO₄ Synthesized by Microwave Induced Plasma Method. *Russ. J. Inorg. Chem.* **2018**, *63*, 725–731. [[CrossRef](#)]
119. Sofronov, D.; Sofronova, E.; Starikov, V.; Baymer, V.; Kudin, K.; Matejchenko, P.; Mamalis, A.; Lavrynenko, S. Microwave Synthesis of Tetragonal Phase CdWO₄. *Mater. Manuf. Process.* **2012**, *27*, 490–493. [[CrossRef](#)]
120. Zhai, Y.; Zhao, X.; Liu, C.; Song, P.; Jing, X.; Han, Y.; Wang, J. CaMoO₄:Dy³⁺, Eu³⁺ phosphors: Microwave synthesis, characterization, tunable luminescence properties and energy transfer mechanism. *Optik* **2018**, *164*, 433–442. [[CrossRef](#)]
121. Perera, S.S.; Munasinghe, H.N.; Yatooma, E.N.; Rabuffetti, F.A. Microwave-assisted solid-state synthesis of NaRE(MO₄)₂ phosphors (RE = La, Pr, Eu, Dy; M = Mo, W). *Dalt. Trans.* **2020**, *49*, 7914–7919. [[CrossRef](#)] [[PubMed](#)]
122. Du, J.; Poelman, D. Facile synthesis of mn⁴⁺-activated double perovskite germanate phosphors with near-infrared persistent luminescence. *Nanomaterials* **2019**, *9*, 1759. [[CrossRef](#)] [[PubMed](#)]

123. Chen, Z.; Yan, Y.; Liu, J.; Yin, Y.; Wen, H.; Zao, J.; Liu, D.; Tian, H.; Zhang, C.; Li, S. Microwave induced solution combustion synthesis of nano-sized phosphors. *J. Alloys Compd.* **2009**, *473*, 13–16. [[CrossRef](#)]
124. Barros, B.S.; De Oliveira, R.S.; Kulesza, J.; Melo, V.R.M.; Melo, D.M.A.; Alves, S. Reddish-orange $\text{Ca}_{3-x}\text{Al}_2\text{O}_6:x\text{Eu}^{3+}$ nanophosphors: Fast synthesis and photophysical properties. *J. Phys. Chem. Solids* **2015**, *78*, 90–94. [[CrossRef](#)]
125. Zhang, P.; Lingxia, L.; Yuming, T. The low temperature synthesis of Eu^{2+} and Dy^{3+} activated $\text{Sr}_3\text{Al}_2\text{O}_6$ nanophosphors by microwave method. *Phys. B Condens. Matter* **2009**, *404*, 4286–4289. [[CrossRef](#)]
126. Lau, K.S.; Hassan, Z.; Lim, W.F.; Mohammad, S.M.; Quah, H.J. Effect of microwave time on the structural and luminescence properties of YAG:Ce prepared by microwave solution combustion (MSC) synthesis. *Optik* **2020**, *212*, 164437. [[CrossRef](#)]
127. Anand, G.T.; Kennedy, L.J. One-pot microwave combustion synthesis of porous $\text{Zn}_{1-x}\text{Cu}_x\text{Al}_2\text{O}_4$ ($0 \leq x \leq 0.5$) Spinel Nanostructures. *J. Nanosci. Nanotechnol.* **2013**, *13*, 3096–3103. [[CrossRef](#)]
128. Tong, Y.-Y.; Li, Y.-F.; Chen, R.; Yang, R.; Sun, L.-L.; Zhang, S.; Fu, Y.-Y. Luminescence characteristics and Förster resonance energy transfer of Y and Bi codoped $\text{Al}_6\text{Si}_2\text{O}_{13}:\text{Eu}$ red nanophosphors obtained by an energy-saving microwave-assisted synthesis. *J. Lumin.* **2020**, *224*, 117234. [[CrossRef](#)]
129. Parhi, P.; Manivannan, V. Novel microwave assisted solid state metathesis synthesis of KMF_3 ($\text{M} = \text{Zn}, \text{Mn}, \text{Mg}, \text{and Co}$). *Mater. Lett.* **2008**, *62*, 3468–3470. [[CrossRef](#)]
130. Guzzetta, F.; Roig, A.; Julián-López, B. Ultrafast Synthesis and Coating of High-Quality $\beta\text{-NaYF}_4:\text{Yb}^{3+}, \text{Ln}^{3+}$ Short Nanorods. *J. Phys. Chem. Lett.* **2017**, *8*, 5730–5735. [[CrossRef](#)] [[PubMed](#)]
131. Makhov, V.N.; Vanetsev, A.S.; Khaidukov, N.M.; Yin, M.; Wei, X.T.; Kotlov, A.; Belsky, A.N. Intrinsic and impurity luminescence of rare earth ions doped KYF_4 nanophosphors. *Radiat. Meas.* **2013**, *56*, 393–396. [[CrossRef](#)]
132. Yang, M.; Deng, G.; Hou, T.; Jia, X.; Wang, Y.; Wang, Q.; Li, B.; Liu, J.; Liu, X. Facile microwave-assisted synthesis of $\text{Zn}_2\text{GeO}_4:\text{Mn}^{2+}, \text{Yb}^{3+}$ uniform nanorods and near-infrared down-conversion properties. *Opt. Mater.* **2017**, *64*, 152–159. [[CrossRef](#)]
133. Lim, C.S.; Atuchin, V.V.; Aleksandrovsky, A.S.; Molokeev, M.S.; Oreshonkov, A.S. Incommensurately modulated structure and spectroscopic properties of $\text{CaGd}_2(\text{MoO}_4)_4:\text{Ho}^{3+}/\text{Yb}^{3+}$ phosphors for up-conversion applications. *J. Alloys Compd.* **2017**, *695*, 737–746. [[CrossRef](#)]
134. Liu, J.; Liu, K.; Gao, H.; Liu, Y.; Yang, C.; Liu, Z. Synthesis and luminescence properties of $\text{CaMoO}_4:\text{Er}^{3+}/\text{Yb}^{3+}$ nanoparticles. *J. Mater. Sci. Mater. Electron.* **2015**, *26*, 3380–3383. [[CrossRef](#)]
135. Liu, X.; Lv, L.; Huang, S.; Su, Y.; Wang, X. Microwave-assisted synthesis and luminescence properties of $\text{Cd}_{1-x}\text{Eu}_x\text{MoO}_4$ red phosphor. *J. Nanosci. Nanotechnol.* **2014**, *14*, 3618–3622. [[CrossRef](#)] [[PubMed](#)]
136. Çelik Gül, G.; Kurtuluş, F. Cr and Co doped $\text{Cs}_5\text{Bi}(\text{MoO}_4)_4$: Microwave assisted synthesis, characterization and specification of optical properties. *Optik* **2017**, *132*, 153–163. [[CrossRef](#)]
137. Huang, R.; Wang, Q.; Liao, J.; Weixiong, Y.O.U. Microwave hydrothermal synthesis and upconversion luminescence properties of $\text{Yb}^{3+}/\text{Tm}^{3+}$ co-doped $\text{NaY}(\text{MoO}_4)_2$ phosphor. *Bull. Mater. Sci.* **2017**, *40*, 1447–1453. [[CrossRef](#)]
138. Lim, C.S. Microwave sol-gel derived $\text{Ho}^{3+}/\text{Yb}^{3+}$ co-doped $\text{NaCaGd}(\text{MoO}_4)_3$ phosphors and their upconversion photoluminescence. *J. Korean Ceram. Soc.* **2016**, *53*, 456–462. [[CrossRef](#)]
139. Lim, C.S. Synthesis of $\text{NaCaLa}(\text{MoO}_4)_3:\text{Ho}^{3+}/\text{Yb}^{3+}$ phosphors via microwave Sol-Gel route and their upconversion photoluminescence properties. *Korean J. Mater. Res.* **2016**, *26*, 363–369. [[CrossRef](#)]
140. Lim, C.S. Microwave sol-gel derived $\text{NaCaGd}(\text{MoO}_4)_3:\text{Er}^{3+}/\text{Yb}^{3+}$ phosphors and their upconversion photoluminescence properties. *Infrared Phys. Technol.* **2016**, *76*, 353–359. [[CrossRef](#)]
141. Li, Y.; Liu, X. Photoluminescence properties of $\text{NaY}(\text{MoO}_4)_2:\text{Eu}^{3+}$ phosphor synthesized by microwave assisted hydrothermal method. *Mater. Sci. Eng. B Solid-State Mater. Adv. Technol.* **2014**, *188*, 20–25. [[CrossRef](#)]
142. Lim, C.S. Highly modulated structure and upconversion photoluminescence properties of $\text{PbGd}_2(\text{MoO}_4)_4:\text{Er}^{3+}/\text{Yb}^{3+}$ phosphors. *Mater. Res. Bull.* **2016**, *75*, 211–216. [[CrossRef](#)]
143. Chou, W.C.; Chang, C.Y.; Huang, S.C.; Hsieh, T.L.; Chung, S.L. Microwave synthesis of red-emitting $\text{Ca}_2\text{Si}_5\text{N}_8:\text{Eu}^{2+}$ phosphor and its photoluminescence properties for white-LEDs. *J. Alloys Compd.* **2019**, *782*, 747–753. [[CrossRef](#)]
144. Chen, W.; Liu, M.; Liu, Y.; Lin, Y.; Yu, L.; Hong, J. Synthesis of luminescent $\text{SnO}_2:\text{Eu}^{3+}$ nanorods via a salt-assisted solution combustion process. *J. Nanosci. Nanotechnol.* **2014**, *14*, 5529–5534. [[CrossRef](#)] [[PubMed](#)]
145. Dominguez, R.D.; Alarcón-Flores, G.; Aguilar-Frutis, M.; Sánchez-Alarcón, R.I.; Falcony, C.; Dorantes-Rosales, H.J.; González-Velázquez, J.L.; Rivas-López, D.I. Effect on the stabilization of the anatase phase and luminescent properties of samarium-doped TiO_2 nanocrystals prepared by microwave irradiation. *J. Alloys Compd.* **2016**, *687*, 121–129. [[CrossRef](#)]
146. Kaszewski, J.; Olszewski, J.; Rosowska, J.; Witkowski, B.; Wachnicki, Ł.; Wenelska, K.; Mijowska, E.; Gajewski, Z.; Godlewski, M.; Godlewski, M.M. $\text{HfO}_2:\text{Eu}$ nanoparticles excited by X-rays and UV-visible radiation used in biological imaging. *J. Rare Earths* **2019**, *37*, 1176–1182. [[CrossRef](#)]
147. Rahim, S.; Ayob, M.T.M.; Hasim, M.H.; Abdul Rahman, I.; Radiman, S. Physical and optical studies of $\text{Gd}_2\text{O}_2\text{S}:\text{Eu}^{3+}$ nanophosphors by microwave irradiation and γ -irradiation methods. *Luminescence* **2019**, *34*, 699–706. [[CrossRef](#)]
148. Bühler, G.; Feldmann, C. Microwave-Assisted Synthesis of Luminescent $\text{LaPO}_4:\text{Ce}, \text{Tb}$ Nanocrystals in Ionic Liquids. *Angew. Chem. Int. Ed.* **2006**, *45*, 4864–4867. [[CrossRef](#)]
149. Li, W.; Lee, J. Microwave-assisted sol-gel synthesis and photoluminescence characterization of $\text{LaPO}_4:\text{Eu}^{3+}, \text{Li}^+$ nanophosphors. *J. Phys. Chem. C* **2008**, *112*, 11679–11684. [[CrossRef](#)]

150. Wang, X.; Wang, Z.; Huang, S.; Shi, X.; Li, J.G. Luminescence responses of CePO₄: Tb monospheres toward vitamin C and permanganate. *Mater. Res. Bull.* **2019**, *117*, 28–34. [[CrossRef](#)]
151. Mokoena, P.P.; Gohain, M.; Bezuidenhout, B.C.B.; Swart, H.C.; Ntwaaborwa, O.M. Luminescent properties and particle morphology of Ca₃(PO₄)₂:Gd³⁺,Pr³⁺ phosphor powder prepared by microwave assisted synthesis. *J. Lumin.* **2014**, *155*, 288–292. [[CrossRef](#)]
152. Rodriguez-Liviano, S.; Aparicio, F.J.; Rojas, T.C.; Hungria, A.B.; Chinchilla, L.E.; Ocaña, M. Microwave-Assisted synthesis and luminescence of mesoporous RE-Doped YPO₄ (RE = Eu, Ce, Tb, and Ce + Tb) nanophosphors with lenticular shape. *Cryst. Growth Des.* **2012**, *12*, 635–645. [[CrossRef](#)]
153. Wang, L.; Xu, M.; Sheng, R.; Liu, L.; Jia, D. Microwave assisted co-precipitation synthesis and photoluminescence characterization of spherical Sr₂P₂O₇:Ce³⁺,Tb³⁺ phosphors. *J. Alloys Compd.* **2013**, *579*, 343–347. [[CrossRef](#)]
154. Zhang, Y.; Xu, J.; Cui, Q.; Yang, B. Eu³⁺-doped Bi₄Si₃O₁₂ red phosphor for solid state lighting: Microwave synthesis, characterization, photoluminescence properties and thermal quenching mechanisms. *Sci. Rep.* **2017**, *7*, 1–12. [[CrossRef](#)] [[PubMed](#)]
155. Birkel, A.; Darago, L.E.; Morrison, A.; Lory, L.; George, N.C.; Mikhailovsky, A.A.; Birkel, C.S.; Seshadri, R. Microwave assisted preparation of Eu²⁺-doped Åkermanite Ca₂MgSi₂O₇. *Solid State Sci.* **2012**, *14*, 739–745. [[CrossRef](#)]
156. Yang, H.; Shi, J.; Gong, M.; Cheah, K.W. Synthesis and photoluminescence of Eu³⁺- or Tb³⁺-doped Mg₂SiO₄ nanoparticles prepared by a combined novel approach. *J. Lumin.* **2006**, *118*, 257–264. [[CrossRef](#)]
157. Yang, H.; Gong, M.; Liang, H. Fine particle Sr₂SiO₄:Tb phosphor for the plasma display panel prepared by a combined approach. *Mater. Res. Bull.* **2010**, *45*, 805–808. [[CrossRef](#)]
158. Sun, J.; Mi, X.; Ma, J.; Su, J.; Xia, W.; Zhang, L.; Bai, Z.; Zhang, X. Rapid synthesis and luminescence properties of Sr₃SiO₅:Eu²⁺ phosphors. *Funct. Mater. Lett.* **2016**, *9*, 1–5. [[CrossRef](#)]
159. Cadiş, A.I.; Muresan, L.E.; Perhaita, I.; Popovici, E.J.; Barbu-Tudoran, L.; Indrea, E. Studies regarding ZnS:Mn nanopowders prepared from single source molecular precursor using microwave-assisted decomposition. *Mater. Res. Bull.* **2016**. [[CrossRef](#)]
160. Arora, S.; Manoharan, S.S. Structural and photo luminescent properties of uncapped nanocrystalline Cd_{1-x}Zn_xS solid solutions. *Opt. Mater.* **2008**, *31*, 176–180. [[CrossRef](#)]
161. Wang, H.; Lu, J.; Wang, R.; Dong, Y.; Ding, L. Synthesis and characterization of the CaTiO₃:Eu³⁺ red phosphor by an optimized microwave-assisted sintering process. *Materials* **2020**, *13*, 874. [[CrossRef](#)]
162. Liao, J.; Wang, Q.; Lan, L.; Guo, J.; Nie, L.; Liu, S.; Wen, H.R. Microwave hydrothermal synthesis and temperature sensing behavior of Lu₂Ti₂O₇:Yb³⁺/Er³⁺ nanophosphors. *Curr. Appl. Phys.* **2017**, *17*, 427–432. [[CrossRef](#)]
163. Yun, H.-S.; Yun, B.-G.; Lee, H.-M.; Jeong, D.-Y.; Lee, W.-I.; Cho, N.-H. Low-temperature synthesis of nanoscale BaTiO₃ powders via microwave-assisted solid-state reaction. *SN Appl. Sci.* **2019**, *1*, 1366. [[CrossRef](#)]
164. Lim, C.S.; Aleksandrovsky, A.; Molokeyev, M.; Oreshonkov, A.; Atuchin, V. Microwave sol-gel synthesis and upconversion photoluminescence properties of CaGd₂(WO₄)₄:Er³⁺/Yb³⁺ phosphors with incommensurately modulated structure. *J. Solid State Chem.* **2015**, *228*, 160–166. [[CrossRef](#)]
165. Wei, B.; Liu, Z.; Xie, C.; Yang, S.; Tang, W.; Gu, A.; Wong, W.T.; Wong, K.L. Fast synthesis of red Li₃BaSrLn₃(WO₄)₈:Eu³⁺ phosphors for white LEDs under near-UV excitation by a microwave-assisted solid state reaction method and photoluminescence studies. *J. Mater. Chem. C* **2015**, *3*, 12322–12327. [[CrossRef](#)]
166. Fan, M.; Liu, S.; Yang, K.; Guo, J.; Wang, J.; Wang, X.; Liu, Q.; Wei, B. Fast synthesis and energy transfer of the tunable single-phase white-emitting phosphor Li₂Gd₄(WO₄)₇:Dy³⁺, Tm³⁺ for WLEDs. *Ceram. Int.* **2020**, *46*, 6926–6933. [[CrossRef](#)]
167. Xu, H.; Xu, K.; Lu, A.; Wang, X.; Hu, J. Microwave hydrothermal synthesis and white up-conversion emission of NaGd(WO₄)₂:(Yb³⁺/Tm³⁺/Ho³⁺) phosphors. *J. Mater. Sci. Mater. Electron.* **2015**, *26*, 3921–3925. [[CrossRef](#)]
168. Yang, Y.; Feng, H.; Zhang, X. Microwave heating synthesis and luminescence of NaY(WO₄)₂:(Ho³⁺, Yb³⁺) phosphors. *J. Mater. Sci. Mater. Electron.* **2015**, *26*, 229–233. [[CrossRef](#)]
169. Liu, Q.; Guo, J.; Fan, M.; Zhang, Q.; Liu, S.; Wong, K.L.; Liu, Z.; Wei, B. Fast synthesis of Dy³⁺ and Tm³⁺ co-doped double perovskite NaLaMgWO₆: A thermally stable single-phase white-emitting phosphor for WLEDs. *J. Mater. Chem. C* **2020**, *8*, 2117–2122. [[CrossRef](#)]
170. Mishra, S.; Priyadarshinee, M.; Debnath, A.K.; Muthe, K.P.; Mallick, B.C.; Das, N.; Parhi, P. Rapid microwave assisted hydrothermal synthesis cerium vanadate nanoparticle and its photocatalytic and antibacterial studies. *J. Phys. Chem. Solids* **2020**, *137*, 109211. [[CrossRef](#)]
171. Jin, Y.; Li, C.; Xu, Z.; Cheng, Z.; Wang, W.; Li, G.; Lin, J. Microwave-assisted hydrothermal synthesis and multicolor tuning luminescence of YP_xV_{1-x}O₄:Ln³⁺ (Ln = Eu, Dy, Sm) nanoparticles. *Mater. Chem. Phys.* **2011**, *129*, 418–423. [[CrossRef](#)]
172. Liu, Y.; Xiong, H.; Zhang, N.; Leng, Z.; Li, R.; Gan, S. Microwave synthesis and luminescent properties of YVO₄:Ln³⁺ (Ln = Eu, Dy and Sm) phosphors with different morphologies. *J. Alloys Compd.* **2015**, *653*, 126–134. [[CrossRef](#)]